

# Interplay of charge carrier and antiferromagnetic order in metallic triangular lattice $\text{GdZn}_3\text{P}_3$

Jiesen Guo<sup>1</sup>,<sup>\*</sup> Fan Yang<sup>1</sup>, Yuzhou He,<sup>2</sup> Xinyang Liu,<sup>1,2,3</sup> Zheng Deng,<sup>2,4</sup> Qinghua Zhang,<sup>2,\*</sup> Xiancheng Wang,<sup>2,4</sup> Xianlei Sheng<sup>1,†</sup>, Wei Li<sup>3,5,6</sup>, Changqing Jin,<sup>2,4</sup> and Kan Zhao<sup>1,‡</sup>

<sup>1</sup>*School of Physics, Beihang University, Beijing 100191, China*


<sup>2</sup>*Beijing National Laboratory for Condensed Matter Physics, Institute of Physics, Chinese Academy of Sciences, Beijing 100190, China*

<sup>3</sup>*CAS Key Laboratory of Theoretical Physics, Institute of Theoretical Physics, Chinese Academy of Sciences, Beijing 100190, China*

<sup>4</sup>*School of Physical Sciences, University of Chinese Academy of Sciences, Beijing 1000190, China*

<sup>5</sup>*CAS Center for Excellence in Topological Quantum Computation, University of Chinese Academy of Sciences, Beijing 100190, China*

<sup>6</sup>*Peng Huanwu Collaborative Center for Research and Education, Beihang University, Beijing 100191, China*

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We investigate a hexagonal  $\text{ScAl}_3\text{C}_3$ -type antiferromagnet  $\text{GdZn}_3\text{P}_3$  single crystal. Compared with antiferromagnetic topological material  $\text{EuM}_2\text{X}_2$  ( $M = \text{Zn, Cd}$ ;  $X = \text{P, As}$ ), the  $\text{GdZn}_3\text{P}_3$  features an additional  $\text{ZnP}_3$  trigonal planar units. Notably, single-crystal x-ray diffraction analysis reveals that Zn and P atoms within a trigonal planar layer exhibit significant anisotropic displacement parameters with space group  $P6_3/mmc$ . Meanwhile, a scanning transmission electron microscopy experiment demonstrates the presence of interstitial P atoms above and below the  $\text{ZnP}$  honeycomb lattice, suggesting potential  $\text{ZnP}$  bond instability within the  $\text{ZnP}_3$  trigonal layers. Concerning the triangular  $\text{Gd}^{3+}$  layer, the magnetic susceptibility  $\chi(T)$  and heat capacity measurements reveal an A-type antiferromagnetic order at  $T_N = 4.5$  K. Below  $T_N$ , the in-plane  $\chi(T)$  is nearly 4 times the  $\chi(T)$  along  $c$  axis, indicative of strong magnetic anisotropy for salt-flux-grown  $\text{GdZn}_3\text{P}_3$  crystal. The density function theory calculation shows that  $\text{GdZn}_3\text{P}_3$  is an indirect semiconductor with a band gap 0.27 eV, supported by the resistivity measurement on the polycrystal sample. Interestingly, Hall measurements of metallic  $\text{GdZn}_3\text{P}_3$  crystals grown via salt flux reveal a low concentration of hole carriers, consistent with the previously mentioned presence of interstitial P atoms. Furthermore, these crystals exhibit a pronounced nonlinear anomalous Hall effect below  $T_N$ . In contrast,  $\text{GdZn}_3\text{P}_3$  crystals grown by chemical vapor transport display weak semiconducting behavior. Despite exhibiting a similar  $T_N$ , the in-plane  $\chi(T)$  below  $T_N$  in the vapor-transport-grown crystal is approximately half in magnitude compared to that of the salt-flux-grown sample, which supports the hypothesis of charge carrier-mediated Ruderman-Kittel-Kasuya-Yosida interactions. These findings collectively establish  $\text{GdZn}_3\text{P}_3$  as a compelling model system for investigating the coupling between charge carriers and triangular lattice magnetism within a two-dimensional framework.

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## I. INTRODUCTION

Geometric frustration in magnetism has long been a prominent research focus in condensed matter physics, as this model prevents antiparallel spin alignment in the antiferromagnetic ground state. This results in significant entanglement and fluctuations, giving rise to many unique states, such as spin liquid [1–8], spin ice [9–16], and spin supersolid [17–20]. As the prototype model of the two-dimensional (2D) triangular lattice formed by rare earth ions has attracted more attention, notable examples include the quantum spin liquid candidate  $\text{YbMgGaO}_4$  [3–8] and  $\text{ARCh}_2$  ( $A = \text{alkali or monovalent ions}$ ,  $R = \text{rare earth}$ , and  $\text{Ch} = \text{O, S, Se}$ ) [21–24]. While most 2D triangular systems focus on insulators and the presence of itinerant electrons in the triangular lattice material, the

emergence of exotic magnetic order could be driven by the Ruderman-Kittel-Kasuya-Yosida (RKKY) interactions.

Based on the above idea, hexagonal  $\text{ScAl}_3\text{C}_3$ -type  $\text{RM}_3\text{X}_3$  ( $R = \text{rare earth}$ ;  $M = \text{Zn, Cd}$ ;  $X = \text{P, As}$ ) compounds have been widely studied, with  $P6_3/mmc$  space group [25–31]. As shown in Fig. 1(a), the crystal structure of  $\text{RM}_3\text{X}_3$  features alternating layers of  $\text{RX}_6$  octahedra,  $\text{MX}_4$  tetrahedra, and  $\text{MX}_3$  trigonal planar units, and the  $M_{\text{trig}}$  and  $X_{\text{trig}}$  atoms of  $\text{MX}_3$  are cross arranged to form a honeycomb [26]. To compare, the antiferromagnetic topological materials  $\text{EuM}_2\text{X}_2$  ( $M = \text{Zn, Cd}$ ;  $X = \text{P, As}$ ) crystallize in a trigonal system with the  $P\bar{3}m1$  (No. 164) space group, constituting a layered structure composed of edge-sharing  $\text{MX}_4$  tetrahedra and  $\text{EuX}_6$  octahedra [32–45]. The interlayer distance between the triangular lattice layer of  $\text{Eu}^{2+}$  and  $\text{Ce}^{3+}$  ions increases from 7.1683 Å in  $\text{EuCd}_2\text{P}_2$  and 7.3278 Å in  $\text{EuCd}_2\text{As}_2$ , to 10.2387 Å in  $\text{CeCd}_3\text{As}_3$ , which enhances the two-dimensional character of the  $\text{RM}_3\text{X}_3$  structure. Previous studies of  $\text{CeCd}_3\text{X}_3$  ( $X = \text{P, As}$ ) have demonstrated strong easy-plane magnetic anisotropy explained by crystalline electric field (CEF) model [46], which demonstrated the low-carrier-density metallic state,

\*Contact author: [zqh@iphy.ac.cn](mailto:zqh@iphy.ac.cn)

†Contact author: [xlsheng@buaa.edu.cn](mailto:xlsheng@buaa.edu.cn)

‡Contact author: [kan\\_zhao@buaa.edu.cn](mailto:kan_zhao@buaa.edu.cn)

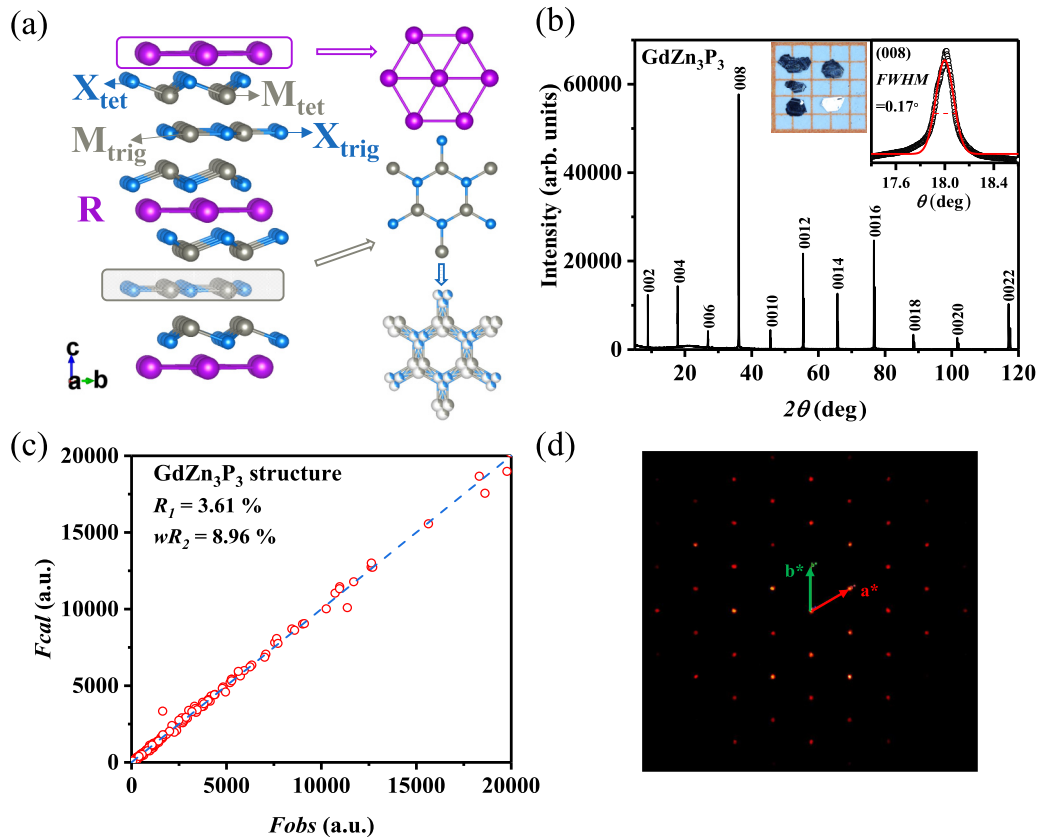


FIG. 1. (a) The crystal structure of  $RM_3X_3$  ( $R$  = rare earth;  $M$  = Zn, Cd;  $X$  = P, As), with space group  $P6_3/mmc$ . Purple, grey, and blue spheres represent  $R$ ,  $M$ , and  $X$  atoms, respectively. The triangular layer formed by  $R$  ions. The honeycomb network formed by  $MX_3$  trigonal planar units, which  $M_{\text{trig}}$  and  $X_{\text{trig}}$  occupy 1, to the local distorted arrangement in which  $M_{\text{trig}}$  and  $X_{\text{trig}}$  are randomly distributed over three equivalent positions, respectively. (b) The XRD pattern of the  $GdZn_3P_3$  (salt) single crystal, with the inset illustrating the rocking curve scan of the (008) peak fitted by Gaussian function, and a photograph of the single crystals also included. (c) Plots of calculated vs experimental structure factors for the refined structure at room temperature. (d) The diffraction spot pattern taken looking along the [001] from the single-crystal x-ray diffraction of  $GdZn_3P_3$  (salt).

with antiferromagnetic (AFM) transitions at about 0.4 K [47,48]. In the  $J_{\text{eff}} = 1/2$  compound  $NdCd_3P_3$ , the AFM transition temperature of 0.34 K is slightly lower than that of  $CeCd_3X_3$  ( $X$  = P, As) [49,50], as well as this trend of magnetic transition being similar to that of  $RZn_3P_3$  ( $R$  = rare earth) [29,51].

During the structural refinement within  $P6_3/mmc$  space group, both  $Cd_{\text{trig}}$  and  $As_{\text{trig}}$  atoms of  $CeCd_3As_3$  exhibit significantly large displacement parameters [26]. As reported in Ref. [26], this pathological issue can be partially removed by splitting atom positions or incorporating twins in a low-symmetry space group. Figure 1(a) illustrates an acceptable model that the  $Cd_{\text{trig}}$  and  $As_{\text{trig}}$  atoms are shifted from the ideal  $2c$  and  $2d$  sites (with  $-6m2$  symmetry) to  $6h$  sites ( $mm2$  symmetry) at a partial occupancy of  $1/3$ . Within the  $6h$  sites, the  $1/3$  occupancy indicates the Cd and As atoms are randomly occupied in the three positions, forming a local distorted arrangement within the  $CdAs_3$  trigonal planar layer. The scenario could be treated as a dynamic movement of  $Cd_{\text{trig}}$  acting on the surrounding  $As_{\text{trig}}$  atoms, which may cause the bond instability of  $(CdAs)_{\text{trig}}$ .

Recently, Ref. [52] provided direct evidence of the local distortion of CdP bonds associated with one-dimensional CdP chains in  $RCd_3P_3$  ( $R$  = La, Ce, Pr, and Nd) single crystals,

based on the observation of diffuse intensity in synchrotron x-ray diffraction (XRD) measurements. By mapping the Cd-P dimer covering problem onto a 2D Ising model defined on the dual triangular lattice, the effective interactions associated with the local bond order on the honeycomb lattice hinder the development of long-range order [52]. Therefore, the coexistence of bond instability and frustrated magnetism renders the  $RCd_3P_3$  system particularly intriguing, as it extends the concept of geometrical frustration from solely encompassing the magnetic component of electrons to incorporating both spin and charge degrees of freedom. It is crucial to identify an isostructural compound with a relatively high magnetic transition temperature to easily investigate the interplay between the Cd/ZnP trigonal planar unit and magnetic triangular  $R^{3+}$  layer.

In this paper, we present the synthesis and characterization of isostructural  $GdZn_3P_3$  single crystal. The ZnP bond instability within  $ZnP_3$  trigonal planar units has been revealed through both single crystal x-ray diffraction (SCXRD) and scanning transmission electron microscopy (STEM) experiments. Based on the magnetic and thermodynamic data, we identify the ground state of  $GdZn_3P_3$  as an A-type AFM order below  $T_N = 4.5$  K. The  $GdZn_3P_3$  system is calculated to be a semiconductor with a band gap of 0.27 eV by density

functional theory (DFT) method. During single-crystal growth, the incorporation of interstitial P atoms leads to hole doping in the  $\text{GdZn}_3\text{P}_3$  system, significantly reducing its resistivity. This effect is evident in crystals grown via chemical vapor transport and ultimately results in metallic conductivity in salt-flux-grown crystals, accompanied by an anomaly at  $T_N$ . Thus, the metallic  $\text{GdZn}_3\text{P}_3$  single crystal gives rise to the unique opportunity of investigating the emergent phenomenon in antiferromagnetic  $\text{RM}_3\text{X}_3$  system, such as the observed nonlinear anomalous Hall effect related with the unconventional spin texture.

## II. EXPERIMENTAL

The polycrystals of  $\text{CeCd}_3\text{As}_3$  and  $\text{GdZn}_3\text{P}_3$  were synthesized by the solid-state method. The CdAs, CdP, and ZnP precursors were first weighed in a glove box using a stoichiometric ratio of Cd powder (99.99%), As lump (99.99%), P lump (99.99%) and Zn powder (99.99%), then mixed in an agate mortar and sealed in an evacuated quartz tube. CdAs was sintered at 500 °C for 30 h, and CdP and ZnP were sintered at 250 °C for 30 h. CdAs and Ce (99.99%), CdP and Gd (99.99%), and ZnP and Gd were mixed at stoichiometric ratios and sintered. As shown in Figs. S1 and S2 in the Supplemental Material [53] (see also Ref. [54] therein), the phase pure  $\text{CeCd}_3\text{As}_3$  and  $\text{GdZn}_3\text{P}_3$  polycrystals have been successfully obtained under sintering at 700 °C for 30 h and 1100 °C for 30 h. Further details are provided in Supplemental Material Note 1 [53]. For transport measurements, the polycrystalline samples were pressed into pellets and subsequently sintered at 800 °C for 15 h.

High-quality single crystals of  $\text{GdZn}_3\text{P}_3$  were grown by 0.5 g of stoichiometric amounts of Gd lump (99.99%), Zn powder (99.99%), and P lump (99.99%) together with 4 g of the NaCl/KCl (1:1) flux [25,29]. The sample was annealed for 24 h at 500 °C, followed by 170 h at 900 °C, then furnace cooled to room temperature. The platelike  $\text{GdZn}_3\text{P}_3$  (salt) single crystals were collected by washing with deionized water.

High-quality single crystals of  $\text{GdZn}_3\text{P}_3$  were also grown by the chemical vapor transport (CVT) method [30]. A mixture of stoichiometric amounts of Gd powder (99.99%), Zn powder (99.99%), and P powder (99.99%) totaling 0.5 g, together with 0.1 g of  $\text{I}_2$  as the transport agent, was sealed in an evacuated quartz tube and placed in a two-zone horizontal furnace. The source and growth zones were maintained at 800 °C and 700 °C for 170 h. After furnace cooling to room temperature, hexagonal  $\text{GdZn}_3\text{P}_3$  (CVT) single crystals were obtained as shown in Fig. S4.

XRD patterns were measured using a Bruker D8 ADVANCE diffractometer with  $\text{Cu K}\alpha$  ( $\lambda = 1.5418 \text{ \AA}$ ) radiation at room temperature. The SCXRD data were performed using a Rigaku XtaLAB Synergy diffractometer at room temperature with the  $\text{Mo K}\alpha$  ( $\lambda = 0.71073 \text{ \AA}$ ) radiation. The single crystal structure solution was completed using Olex2 [55] and SHELXL [56]. The high-angle annular-dark-field (HAADF) image and annular bright-field image were characterized using an ARM-200F (JEOL, Tokyo, Japan) scanning transmission electron microscope (STEM) operated at 200 keV with a CEOS Cs corrector (CEOS GmbH, Heidelberg, Germany) to cope with the probe-forming objective spherical

TABLE I. Crystal data and structure refinement for single-crystal  $\text{GdZn}_3\text{P}_3$  (salt).

Formula	$\text{GdZn}_3\text{P}_3$
T(K)	293(2)
Crystal system	Hexagonal
Space group	$P6_3/mmc$
a(Å)	4.0107(3)
b(Å)	4.0107(3)
c(Å)	19.8808(15)
V(Å) <sup>3</sup>	276.95(4)
$\theta$	2.049–28.158
No. reflections collected	1066
No. of variables	14
Final $R$ indices	$R_1 = 0.0361$ , $wR_2 = 0.0896$
$R$ indices (all data)	$R_1 = 0.0405$ , $wR_2 = 0.0911$
Goodness of fit	0.992
Largest diff. peak and hole ( $\text{e}\text{\AA}^{-3}$ )	0.92, $-1.09$

aberration. The magnetic properties were measured using the Quantum Design Magnetic Property Measurement System (MPMS). The specific heat was measured using the Quantum Design Physical Property Measurement System (PPMS). The resistivity and Hall data was measured using the Cryogen Free Measurement System.

Our DFT uses the generalized gradient approximation (GGA) [57,58] in the form of PBE functional [59] and included spin-orbit coupling, as implemented in the Vienna ab initio Simulation Package [60–62]. The energy cutoff of the plane-wave basis is set to 350 eV. The energy convergence criterion in the self-consistent calculations is set to  $10^{-6}$  eV. A  $9 \times 9 \times 3$   $\Gamma$ -centered Monkhorst-Pack k-point mesh is used for the first Brillouin zone sampling. To account for the correlation effects for Gd, we adopted the GGA +  $U$  method with the value of  $U = 1$  eV.

## III. RESULTS AND DISCUSSION

### A. Crystal structure of $\text{GdZn}_3\text{P}_3$ (salt) single crystal:

#### X-ray diffraction and STEM analysis

A series of (0, 0, 2L) peaks are observed by XRD diffraction experiment in Fig. 1(b), indicating the  $ab$  plane as the natural cleavage plane. Rocking curve analysis of the Bragg peak (008) in the inset of Fig. 1(b) demonstrates a narrow full width at half maximum of  $0.17^\circ$ , indicating high quality of the  $\text{GdZn}_3\text{P}_3$  crystals. The single crystal diffraction spots along [001] in Fig. 1(d) show direct evidence for the six-degree symmetry of hexagonal  $\text{GdZn}_3\text{P}_3$  system, with related spots along [100] and [010] shown in Fig. S5 [53]. The Rietveld refinements of the SCXRD data converge to the agreement factors  $R_1 = 3.61\%$  and  $wR_2 = 8.96\%$  in Fig. 1(c), with  $a = b = 4.0107(3) \text{ \AA}$  and  $c = 19.8808(15) \text{ \AA}$  in Table I. The corresponding crystal structure is depicted in Fig. 2(c), with the refined atomic parameters listed in Table II.

Due to the presence of large voids above and below the location in Fig. 2(c),  $\text{Zn}_{\text{trig}}$  atom exhibits large anisotropic displacement parameters, with  $U_{33} = 0.048$  along the  $c$  axis and  $U_{11} = 0.032$  in the  $ab$  plane at room temperature. These

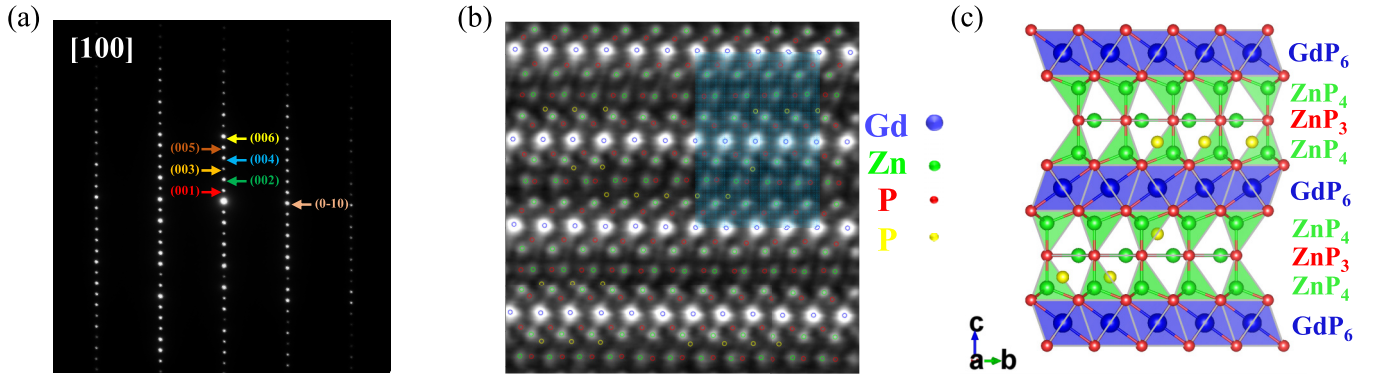


FIG. 2. (a) The electron diffraction pattern along [100] direction. (b) The high-angle annular dark-field (HAADF) image of  $\text{GdZn}_3\text{P}_3$  (salt) taken along [100] direction. The blue, green, red, and yellow spheres represent Gd, Zn, P, and the interstitial impurity P, respectively. (c) Corresponding crystal structure of  $\text{GdZn}_3\text{P}_3$  (salt) along [100] direction, matching the blue-boxed region in Fig. 2(b). The structure features alternating layers of  $\text{GdP}_6$  octahedra,  $\text{ZnP}_4$  tetrahedra, and  $\text{ZnP}_3$  trigonal planar units, with the yellow ball marks the anticipated site of the interstitial P atom.

values are comparable to the values of 0.0501 and 0.02826 for  $\text{Cd}_{\text{trig}}$  atoms of  $\text{PrCd}_3\text{P}_3$ . On the other hand, the  $\text{P}_{\text{trig}}$  atom has larger in-plane displacement with the  $U_{11}$  parameter as 0.031, also comparable with the value of 0.0392 for  $\text{P}_{\text{trig}}$  atoms of  $\text{PrCd}_3\text{P}_3$  [52].

To reveal the structural details of interlayer atoms in atomic scale, we continue to perform STEM on a  $bc$  plane of  $\text{GdZn}_3\text{P}_3$  about  $4 \times 3 \mu\text{m}^2$  area, with details information in Fig. S6 [53,63]. As shown in Fig. 2(a), the selected area electron diffraction pattern contains a series of diffraction peaks, such as (001), (002), (003), and (004), etc., and (0-10). Moreover, the HAADF image in Fig. 2(b) directly shows the atomic arrangement in the unit cell of  $\text{GdZn}_3\text{P}_3$  projected along the [100] direction, corresponding to the crystal structure in Fig. 2(c). First, the HAADF image confirms the full occupancy of  $\text{Zn}_{\text{trig}}$  and  $\text{P}_{\text{trig}}$  atoms within the  $\text{ZnP}_3$  trigonal planar. We conclude that the underlying bond instabilities of  $(\text{ZnP})_{\text{trig}}$  are local characteristics rather than uniform.

Notably, numerous interstitial P atoms are observed in the HAADF image, as indicated by the yellow spheres in Fig. 2(b). Within the honeycomb lattice of  $\text{Zn}_{\text{trig}}$  and  $\text{P}_{\text{trig}}$  atoms, the bond distance is about 2.32 Å, while the distance between  $\text{Zn}_{\text{trig}}$  and  $\text{P}_{\text{tet}}$  atom is 3.29 Å along the  $c$  axis. Thus, the presence of interstitial P atom is closely related with the large voids above and below the location of  $\text{Zn}_{\text{trig}}$  atoms in Fig. 2(c). The estimated occurrence probability of interstitial P atoms per formula unit (f. u.) is inferred from a statistical analysis of multiple spatial regions during the STEM

measurement. Specifically, approximately 24 interstitial P atoms are observed in 44 formula units in the projection plane along the [100] direction in Fig. 2(b), corresponding to  $\sim 0.55$  per f. u., while 18 and 17 interstitial P atoms are observed in 44 formula units along the [100] direction in Figs. S7(a) and S7(b), corresponding to  $\sim 0.41$  and  $\sim 0.39$  per f. u., respectively. On this basis, the interstitial P atoms projected along [100] direction appear with an average probability of  $\sim 0.45$  per f. u.

By combining the two orthogonal projections along and perpendicular to the [100] direction, an average volumetric probability of approximately 0.2 ( $0.45 \times 0.45$ ) is obtained. Taking into account that the weak image contrast of an interstitial P atom is  $\sim 0.2$  of that of a regular P site, the resulting volumetric density is estimated to be  $\sim 0.04$  ( $0.2 \times 0.2$ ) per f.u. Such a low concentration is well below the detection limit of x-ray diffraction, with no corresponding signatures observed in XRD patterns of Fig. 1. And we will discuss the effect of interstitial P atoms in electronic transport section later.

Recently, the structural instability within ZnP/As honeycomb layer has been revealed in its sister compound  $\text{GdZn}_3\text{As}_3$  [31], which crystallizes in a distorted  $\text{ScAl}_3\text{C}_3$ -type orthorhombic structure with  $Cmcm$  space group, analogous to the case of  $\text{YbAl}_3\text{C}_3$  [64]. It warrants performing the synchrotron x-ray diffraction measurement to investigate the potential existence of short-range bond order existing in  $R\text{Zn}_3\text{P}_3$  ( $R = \text{La}, \text{Ce}, \text{Pr}, \text{Nd}$  and  $\text{Gd}$ ) single crystals in future research.

TABLE II. Wyckoff positions, atomic coordinates, occupancies, isotropic and anisotropic displacement parameters ( $\text{\AA}^2$ ), where  $U_{23} = U_{13} = 0$  for  $\text{GdZn}_3\text{P}_3$  (salt).

Atom.	Wyck.	$x$	$y$	$z$	Occup	$U(\text{eq})$	$U_{11} (U_{22})$	$U_{33}$	$U_{12}$
Gd	$2a$	0	0	0.5	1	0.0095(5)	0.0075(6)	0.0134(8)	0.0038(3)
$\text{Zn}_{\text{tet}}$	$4f$	-1/3	1/3	0.6299(1)	1	0.0136(7)	0.0139(9)	0.0130(11)	0.0069(5)
$\text{Zn}_{\text{trig}}$	$2d$	-2/3	2/3	3/4	1	0.0374(13)	0.0321(19)	0.048(3)	0.0160(9)
$\text{P}_{\text{tet}}$	$4f$	2/3	1/3	0.4155(3)	1	0.0112(12)	0.0117(17)	0.010(2)	0.0058(8)
$\text{P}_{\text{trig}}$	$2c$	-1/3	1/3	3/4	1	0.024(2)	0.031(4)	0.011(4)	0.0153(18)

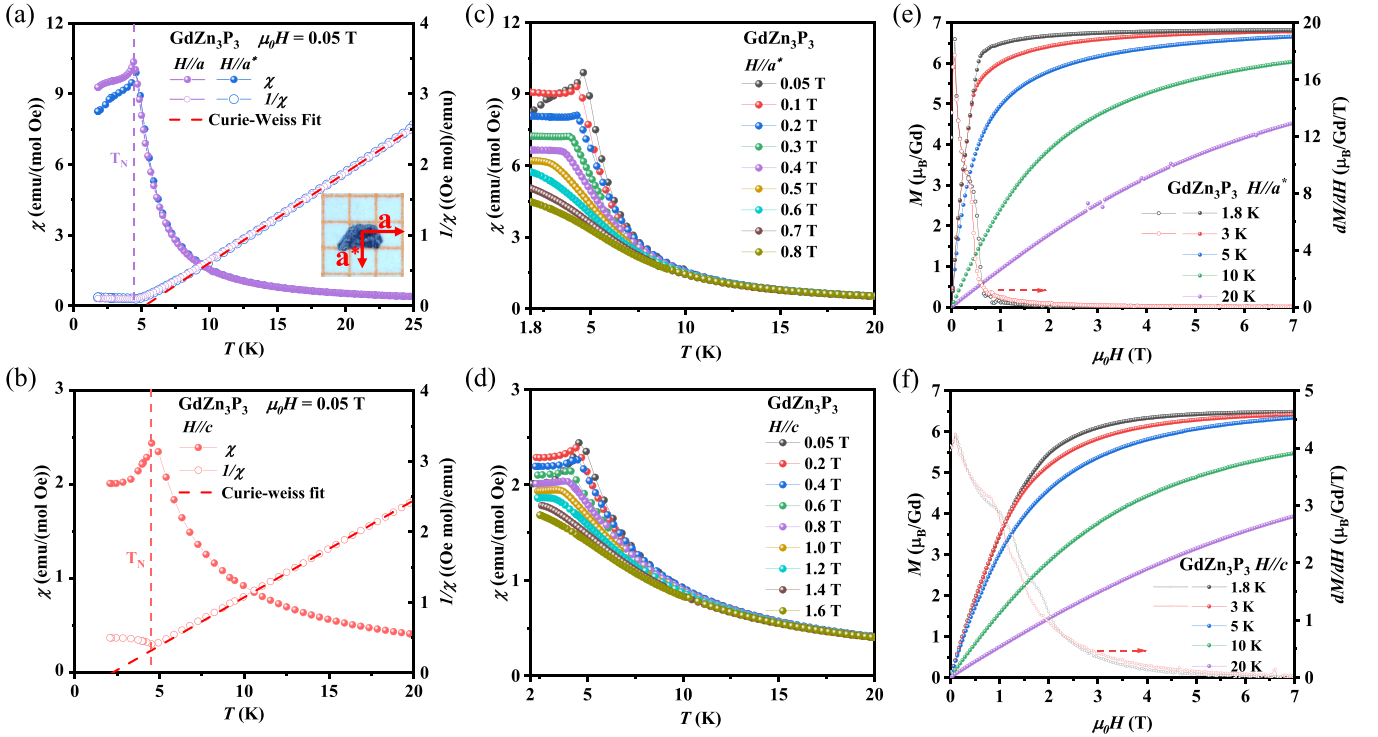


FIG. 3. (a), (b) The temperature-dependent magnetic susceptibility  $\chi(T)$  of  $\text{GdZn}_3\text{P}_3$  (salt) between 1.8 and 25 K measured with magnetic field along  $a$ ,  $a^*$ , and  $c$  directions, an inset optical image of the single crystal. (c), (d) The  $\chi(T)$  below 20 K at various fields along  $a^*$  and  $c$  directions. (e), (f) Isothermal magnetization  $M(H)$  and the differential susceptibility  $dM/dH$  at different temperatures with the magnetic field applied along the  $a^*$  and  $c$  directions.

### B. Magnetic properties and magnetic structure of $\text{GdZn}_3\text{P}_3$ (salt) single crystal

Magnetic susceptibility  $\chi(T)$  measurements of  $\text{GdZn}_3\text{P}_3$  have been performed under magnetic field along  $a$ ,  $a^*$ , and  $c$  axes, showing a clear AFM transition at  $T_N = 4.5$  K, as shown in Figs. 3(a) and 3(b) and Figs. S8–S9 [53]. The  $\chi(T)$  curve exhibits no significant difference between  $a$  and  $a^*$  directions, with its value being about 4 times that along the  $c$  axis below  $T_N$ . In Fig. S9(a), the Curie-Weiss law is used to fit the  $\chi(T)$  (200–300 K) under 0.05 T, yielding  $\mu_{\text{eff}} = 7.93 \mu_B/\text{Gd}$  in the  $ab$  plane, consistent with the expected value  $7.94 \mu_B$ . In Figs. 3(a) and 3(b), the Curie-Weiss law is used to fit the  $\chi(T)$  from 15–25 K for  $H \parallel a$  and from 15–20 K for  $H \parallel c$ , which yields  $\theta_{\text{CW}} = 5.2$  K for  $H \parallel a^*$ , and  $\theta_{\text{CW}} = 2.0$  K for  $H \parallel c$ .

These characteristics suggest the formation of an A-type AFM order, likely with the spin alignment of  $\text{Gd}^{3+}$  ions lying in the  $ab$  plane below  $T_N$  and FM layers stacked antiferromagnetically. From the perfect hexagonal structure to the distorted  $\text{ScAl}_3\text{C}_3$ -type case, accompanied by the release of magnetic frustration, the ground state evolves from A-type AFM order with  $T_N = 4.5$  K in  $\text{GdZn}_3\text{P}_3$  to FM order with  $T_c = 6.3$  K in  $\text{GdZn}_3\text{As}_3$  [31].

Similarly, A-type AFM order has also been observed in topological material  $\text{EuM}_2\text{X}_2$  ( $M = \text{Zn}, \text{Cd}; X = \text{P}, \text{As}$ ), with  $T_N$  at about 23 K in  $\text{EuZn}_2\text{P}_2$  [40–42,44,45] and 9.3 K in  $\text{EuCd}_2\text{As}_2$ . The much lower  $T_N$  in  $\text{GdZn}_3\text{P}_3$  is speculated to be largely due to the extra honeycomb nets ZnP layer between triangular lattice of Gd atoms, thus weakening the AFM interaction strength. According to the previous literature, the magnetic moments of  $\text{EuCd}_2\text{As}_2$  and  $\text{EuZn}_2\text{As}_2$  are

determined to be parallel to  $ab$ -plane through resonant elastic x-ray scattering (REXS) and neutron diffraction; and the spins of  $\text{EuZn}_2\text{P}_2$  are canted out of the  $a$ - $a$  plane by an angle of about  $40^\circ \pm 10^\circ$  and aligned along the [100] direction as confirmed by resonant magnetic x-ray diffraction data [32,39,40,42].

We also note a subtle cusp appears in the  $ab$ -plane  $\chi(T)$  curve at 2.4 K [see Fig. 3(a) and Fig. S8], and to capture this feature, Fig. S8(a) displays the temperature derivative of  $\chi(T)$  ( $d\chi/dT$ ) curve. A modulated magnetic order observed in  $\text{GdV}_6\text{Sn}_6$  indicates the presence of an incommensurate magnetic order below 5.2 K and a commensurate magnetic order below 3.8 K, as demonstrated by REXS measurements [65,66]. To clarify the microscopic magnetic structure together with its temperature dependence of  $\text{GdZn}_3\text{P}_3$ , it seems necessary to conduct REXS and neutron scattering measurements in future.

In addition, the  $\chi(T)$  curves at different magnetic fields were measured as shown in Figs. 3(c) and 3(d). With the magnetic field increased to 0.6 T for  $H \parallel a^*$  and 1.4 T for  $H \parallel c$ , the AFM transition peak has been completely suppressed to below 1.8 K for  $H \parallel a^*$  and 2.5 K for  $H \parallel c$ . The corresponding  $d\chi/dT$  curves are presented in Fig. S9 [53]. As shown in Figs. 3(e) and 3(f), the magnetization saturates at  $M_S \sim 6.7 \mu_B$  for  $H \parallel a^*$  and  $M_S \sim 6.5 \mu_B$  for  $H \parallel c$  at 1.8 K, consistent with the expected value  $7 \mu_B$  for  $\text{Gd}^{3+}$  ions. The transition fields are determined from the  $dM/dH$  curves in Fig. 3(e) as  $\mu_0 H_{ab} = 0.6$  T at 1.8 K and 0.5 T at 3 K for  $H \parallel a^*$ , respectively. For a detailed description of the determination process of the metamagnetic transitions under  $H \parallel c$ ,

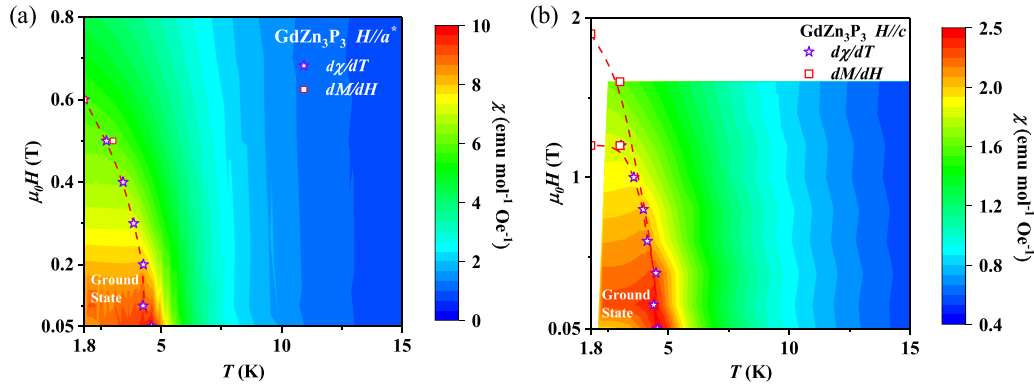


FIG. 4. (a), (b) The  $H$ - $T$  magnetic phase diagram overlaid on the contour plots of the magnetic susceptibility of  $\text{GdZn}_3\text{P}_3$  (salt) with the field along the  $a^*$  and  $c$  directions, respectively. The magnetic phase boundaries were extracted through  $\chi(T)$  and  $M(H)$  measurements. The star and square symbols, respectively, denote the temperature at a specific field where these transition peaks are observed in the temperature derivative of susceptibility  $d\chi/dT$  and the differential susceptibility  $dM/dH$ . The dashed lines represent the magnetic phase boundary.

please refer to Fig. 9(d). Finally, Figs. 4(a) and 4(b) depict the  $H$ - $T$  phase diagram constructed by combining the AFM transition peaks obtained above.

### C. Magnetic heat capacity and entropy of $\text{GdZn}_3\text{P}_3$ (salt) single crystal

Figure 5 presents the temperature dependence of the measured heat capacity  $C_p$  curve, which reveals a sharp transition close to  $T_N = 4.5$  K. The temperature dependence of the specific heat of  $\text{LaZn}_3\text{P}_3$  in zero magnetic field was taken from Ref. [51] and is shown in Fig. 5(a), where the background phonon contribution is fitted by a polynomial expression  $C_{ph} = aT^3 + bT^5 + cT^7 + dT^9 + eT^{11}$  [67,68]. The magnetic heat capacity  $C_m$  is obtained after subtracting phonon contribution. There is still residual magnetic entropy below 1.8 K, so the lower temperature magnetic heat capacity is fitted through the power law  $C_m(T) = b \times T^a$  to get  $a = 1.16$  as shown in Fig. 5(b), which does not follow the  $C_m(T) \sim T^3$  relation observed in the conventional ungapped AFM magnon

excitations [69]. By integrating  $(C_p - C_{ph})/T$  from 15 K to 0 K, the obtained values of the magnetic entropy  $S_{mag}$  are 15.9 J/(mol K), which is 92% of the expected magnetic entropy as  $R \ln 8$ . The heat capacity  $C_p$  data below 1.8 K seems necessary to obtain a more accurate power law fitting as well as the integrated  $S_{mag}$  value.

### D. Electronic band structure DFT calculation and transport results of polycrystalline $\text{GdZn}_3\text{P}_3$

Similar to the  $\text{EuCd}_2\text{As}_2$  case [36],  $\text{GdZn}_3\text{P}_3$  may have three possible A-type AFM structures, as shown in Fig. 6(a), with magnetic moments of  $\text{Gd}^{3+}$  ions along  $b$ ,  $c$ , and  $x$  directions, and the configuration with the moment aligned along the  $b$  axis exhibits the lowest free energy according to DFT calculations. Given the magnetic evidence above, the ground state of  $\text{GdZn}_3\text{P}_3$  would most likely be A-type magnetic structure with Gd moment along the  $b$  axis (i). Figures 6(b) and S11 display the corresponding band structures with spin-orbit coupling included and density of states (DOS) plot near

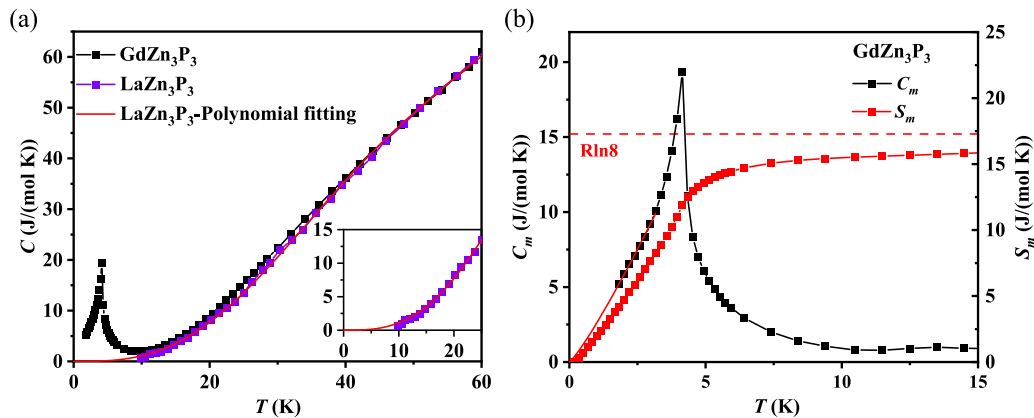


FIG. 5. (a) Temperature-dependent specific heat of  $\text{GdZn}_3\text{P}_3$  (salt) (black square) of  $\text{LaZn}_3\text{P}_3$  (from Ref. [51]) (purple square), The dashed red line represents a polynomial fit temperature-dependent specific heat of  $\text{LaZn}_3\text{P}_3$  to estimate the lattice contribution. The inset shows an enlarged view of the specific heat of  $\text{LaZn}_3\text{P}_3$  polynomial fitting below 25 K and an inset photo of the single crystal of  $\text{GdZn}_3\text{P}_3$  (salt) for heat capacity measurement in the PPMS puck. (b) The  $C_m$  (black line) with the portion of the zero field  $C_m(T)$  fitted to a  $b \times T^a$  power law (red extension line) and temperature-dependent integrated magnetic entropy (red line).

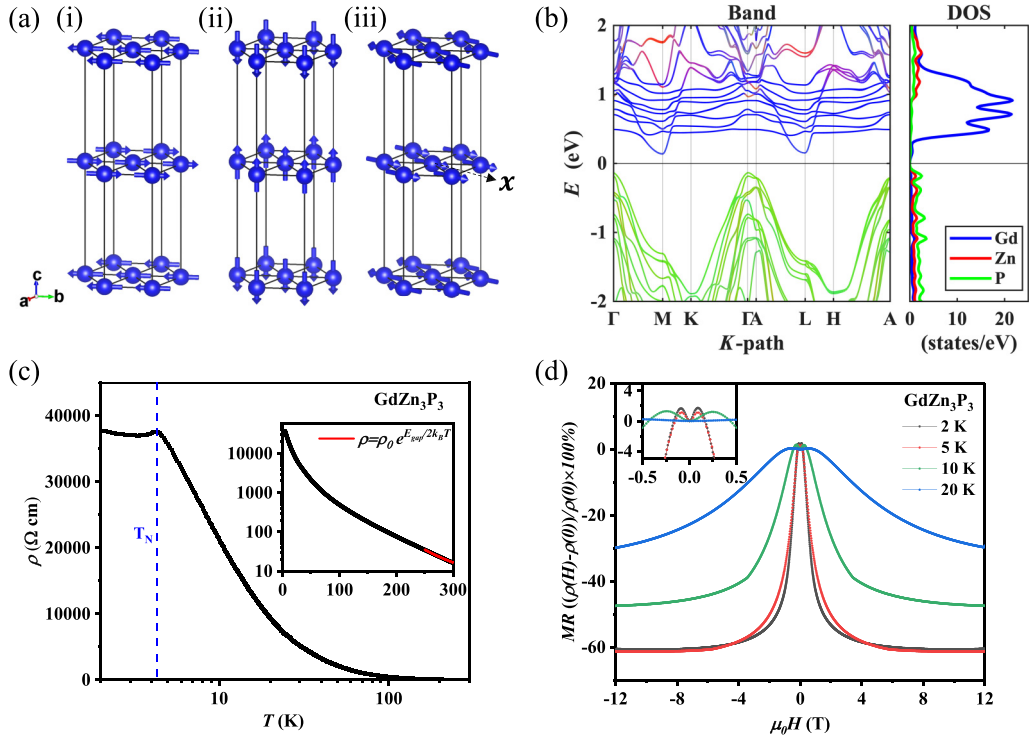


FIG. 6. (a) Proposed possible magnetic structure of  $\text{GdZn}_3\text{P}_3$ . A-type AFM structure configurations on Gd sites with the magnetic moment aligned along  $b$  (i),  $c$  (ii), or  $x$  (iii) directions. (b) (i) corresponding Band structure with spin-orbit coupling included and density of states (DOS) plot near the Fermi level, calculated via density functional theory (DFT). (c) The temperature-dependent resistivity of polycrystalline  $\text{GdZn}_3\text{P}_3$  sample between 2 K and 300 K, with the inset showing the transport activation gap extracted from a fit in the 250–300 K range. (d) Magnetoresistance (MR) of  $\text{GdZn}_3\text{P}_3$  as a function of field at several representative temperatures, with the inset revealing the detailed behavior of MR in the field range of  $-0.5$  T to  $0.5$  T.

the Fermi level, with indirect band gaps from  $\Gamma$  to  $M$  as 0.27 eV, 0.28 eV, and 0.26 eV, respectively [53]. This band gap is smaller than that of polycrystalline  $\text{LaCd}_3\text{P}_3$  (0.75 eV) and  $\text{NdCd}_3\text{P}_3$  (0.63 eV). In addition, the indirect band gap of  $\text{EuZn}_2\text{P}_2$  is calculated to be 0.2 eV, and observed as 0.09 eV in the angle-resolved photoemission spectroscopy experiment [42].

Figure 6(c) illustrates the electrical resistivity of polycrystalline  $\text{GdZn}_3\text{P}_3$  within the temperature range 2–300 K, consistent with the semiconducting behavior observed in  $\text{RCd}_3\text{P}_3$  ( $R = \text{La}, \text{Ce}, \text{and Nd}$ ) polycrystals. As depicted in inset of Fig. 6(c), the transport gap of  $\text{GdZn}_3\text{P}_3$  is extracted from a fit to the resistivity data between 250 and 300 K using the thermally activated Arrhenius form  $\rho = \rho_0 e^{E_{\text{gap}}/2k_B T}$ , yielding a band gap of approximately 0.2 eV, consistent with the DFT calculation results above. Interestingly, the AFM transition also shows up in the resistivity curve, marked as  $T_N = 4.5$  K in Fig. 6(c). A similar characteristic has also been observed in  $\text{EuZn}_2\text{P}_2$  and  $\text{EuCd}_2\text{As}_2$  [21,51].

As shown in inset of Fig. 6(d), the magnetoresistance (MR) curve below 20 K exhibits an initial increase under small field ( $\mu_0 H < 0.3$  T), which may be associated with the impurity carriers—similar behavior also reported in  $\text{EuZn}_2\text{As}_2$  [43]. Under high field, the magnetic scattering would be suppressed in the polarized FM state, thus a negative MR has been observed in polycrystalline  $\text{GdZn}_3\text{P}_3$  under field up to 12 T. According to Figs. 3(e) and 3(f), the  $\text{GdZn}_3\text{P}_3$  compound enters the polarized state under 1 T for  $H \parallel a^*$  and 4 T for

$H \parallel c$ . Thus, the negative MR below 5 K quickly reaches a magnitude of  $-60\%$  under 4 T, then saturates at this value under  $\mu_0 H > 4$  T. The magnitude of negative MR in  $\text{GdZn}_3\text{P}_3$  is compared to that of  $\text{EuZn}_2\text{As}_2$  [43],  $\text{EuZn}_2\text{P}_2$ , and  $\text{EuCd}_2\text{As}_2$  [21,49]. The MR curve becomes more pronounced as temperature decreases from 20 K to 2 K, indicating the significant influence of magnetic ordering behavior on the electronic transport properties in  $\text{GdZn}_3\text{P}_3$ .

### E. Electronic transport of the $\text{GdZn}_3\text{P}_3$ (salt) single crystal

Figure 7(a) illustrates the in-plane resistivity curve of  $\text{GdZn}_3\text{P}_3$  (salt) single crystal, exhibiting metallic behavior at elevated temperatures. Below 50 K, the resistivity slightly increases with cooling, and arising from enhanced magnetic scattering below 10 K, the resistivity rapidly increases to its peak value around  $T_N = 4.5$  K. The resistivity value herein is on the order of 1 m $\Omega$  cm, consistent with the case of  $\text{GdZn}_3\text{As}_3$  single crystal [31].

Figures 7(b) and 7(c) summarize field-dependent MR data from 2 K to 20 K under  $H \parallel ab$  and  $H \parallel c$ , respectively. It is clear that the negative MR along two directions obtains its maximum value as the temperature approaches  $T_N$ . At  $T = 2$  K, the negative MR curve quickly reaches the polarized state under 1 T for  $H \parallel ab$  and 4 T for  $H \parallel c$ , consistent with the transitions observed in the  $M(H)$  curves [Figs. 3(e) and 3(f)]. Both in-plane and out-of-plane MRs reach approximately 40%, which is smaller than that observed in the

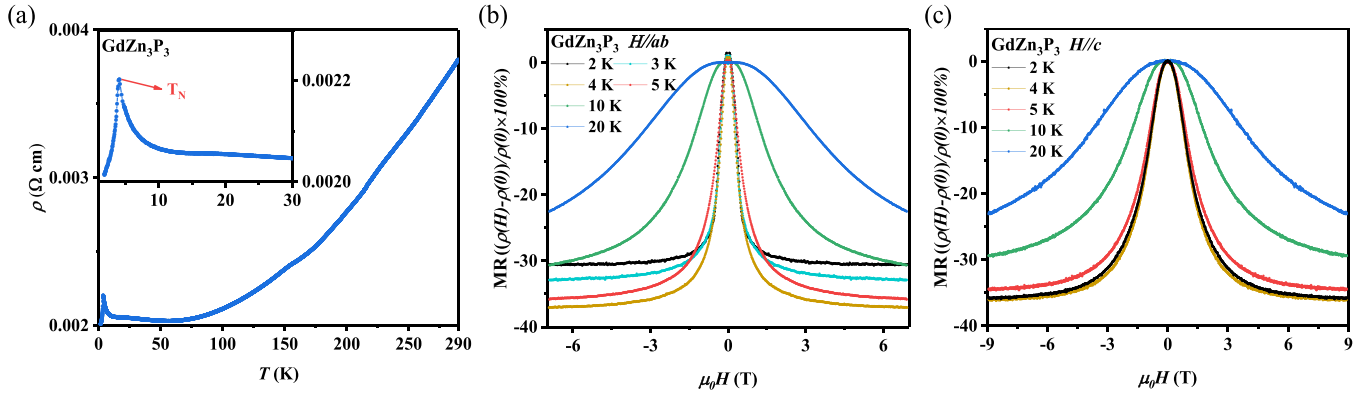


FIG. 7. (a) The temperature-dependent resistivity of single crystal  $\text{GdZn}_3\text{P}_3$  (salt), with the inset revealing the detailed behavior of resistivity below 30 K. (b), (c) Magnetoresistance (MR) of  $\text{GdZn}_3\text{P}_3$  (salt) as a function of field applied within the  $ab$  plane and along the  $c$  axis at several representative temperatures.

polycrystalline sample due to enhanced magnetic and grain-boundary scattering in the latter.

The metallic conductivity of  $\text{GdZn}_3\text{P}_3$  (salt) single crystal enables the Hall measurement down to low temperatures. We note the Hall data presented in Fig. 8 exhibits larger noise to signal ratio compared with the MR curves in Fig. 7 due to the relatively high contact resistance and thin thickness in the salt-flux grown crystals, as mentioned in  $\text{GdZn}_3\text{As}_3$  case as well. In the latter, the Hall data was only obtained down to 10 K with acceptable quality.

At 50 K and 80 K, the Hall resistivity of  $\text{GdZn}_3\text{P}_3$  shows a positive slope as a function of field, indicating hole-type carriers. The carrier concentration, obtained from linear fits to the Hall data, is approximately  $1 \times 10^{19} \text{ cm}^{-3}$  and  $1.3 \times 10^{19} \text{ cm}^{-3}$ , corresponding to  $1.4 \times 10^{-3}$  and  $1.8 \times 10^{-3}$  holes per f.u., comparable to  $\text{CeCd}_3\text{P}_3$  ( $\sim 2 \times 10^{-3}$  holes per f.u.) [47], and  $\text{GdZn}_3\text{As}_3$  ( $\sim 2.3 \times 10^{-3}$  holes per f.u.) [31]. It should be noted that the interstitial P atoms occupy

nonstandard positions within the hexagonal  $\text{ScAl}_3\text{C}_3$ -type crystal structure. Consequently, it is unlikely that all interstitial P atoms with volumetric density roughly 0.04 per f.u. serve as effective dopants contributing free carriers to the band structure of semiconducting  $\text{GdZn}_3\text{P}_3$ . The observed hole-type conductivity in  $\text{GdZn}_3\text{P}_3$  single crystal is therefore attributed to the presence of interstitial P atoms, consistent with the Hall effect measurements presented above.

The Hall resistivity exhibits a nonlinear dependence on the magnetic field at 20 K, and an additional humplike feature appears at  $T = 3 \text{ K}$ , below  $T_N$ . This behavior gradually disappears as  $\text{GdZn}_3\text{P}_3$  approaches saturation state under  $\mu_0 H > 4 \text{ T}$ , which typically originates from nonzero Berry curvature induced by finite scalar spin chirality within magnetic domain walls. This phenomenon, known as the nonlinear AHE, has been observed in topological candidate materials  $\text{EuCd}_2\text{As}_2$  and  $\text{EuZn}_2\text{Sb}_2$  [37,70], as well its sister compound  $\text{GdZn}_3\text{As}_3$  [31].

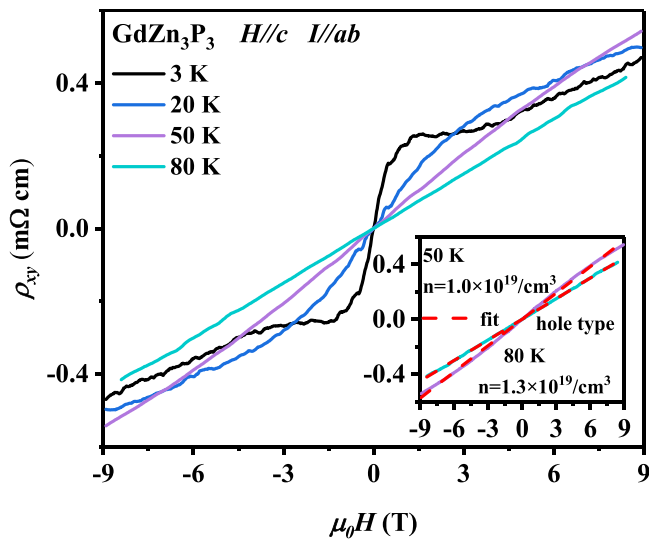


FIG. 8. Magnetic field dependence of the Hall resistivity ( $\rho_{xy}$ ) for the single crystal  $\text{GdZn}_3\text{P}_3$  (salt) at various temperatures (3, 20, 50, and 80 K). The inset displays linear fits to the Hall resistivity data at 50 K and 80 K.

### E. Physical property of the $\text{GdZn}_3\text{P}_3$ (CVT) single crystal

As mentioned above, the resistivity values of  $\text{GdZn}_3\text{P}_3$  (salt) crystal are on the order of 1 m $\Omega$  cm, significantly lower than the 10 m $\Omega$  cm value previously reported for  $\text{GdZn}_3\text{P}_3$  single crystals in Ref. [29]. According to our Hall measurement, the different resistivity value might be caused by the different density of hole type carriers and interstitial P atoms.

To explore the coupling between charge carrier and antiferromagnetism in  $\text{GdZn}_3\text{P}_3$ , we continue to grow  $\text{GdZn}_3\text{P}_3$  crystal with the CVT method, and characterize its physical property below. As depicted in Fig. 9(a), the  $\text{GdZn}_3\text{P}_3$  (CVT) single crystal shows weak semiconducting behavior, with the resistivity value increasing from 27 m $\Omega$  cm at 300 K to 0.7  $\Omega$  cm at 2 K. Despite exhibiting a similar  $T_N$ , the magnitude of  $\chi(T)$  along the  $a$  axis below  $T_N$  in Fig. 9(b) is approximately half that of the  $\text{GdZn}_3\text{P}_3$  (salt) sample, while the value along the  $c$  axis is slightly lower than that of latter crystal. In Fig. 9(b), the Curie-Weiss fitting of  $\chi(T)$  (15–25 K) yields  $\theta_{\text{CW}} = 4.1 \text{ K}$  for  $H \parallel a$  and  $\theta_{\text{CW}} = 1.6 \text{ K}$  for  $H \parallel c$ , which are comparable to the  $\text{GdZn}_3\text{P}_3$  (salt) case. Notably, the anomaly observed in the in-plane  $\chi(T)$  curves of the salt-flux-grown

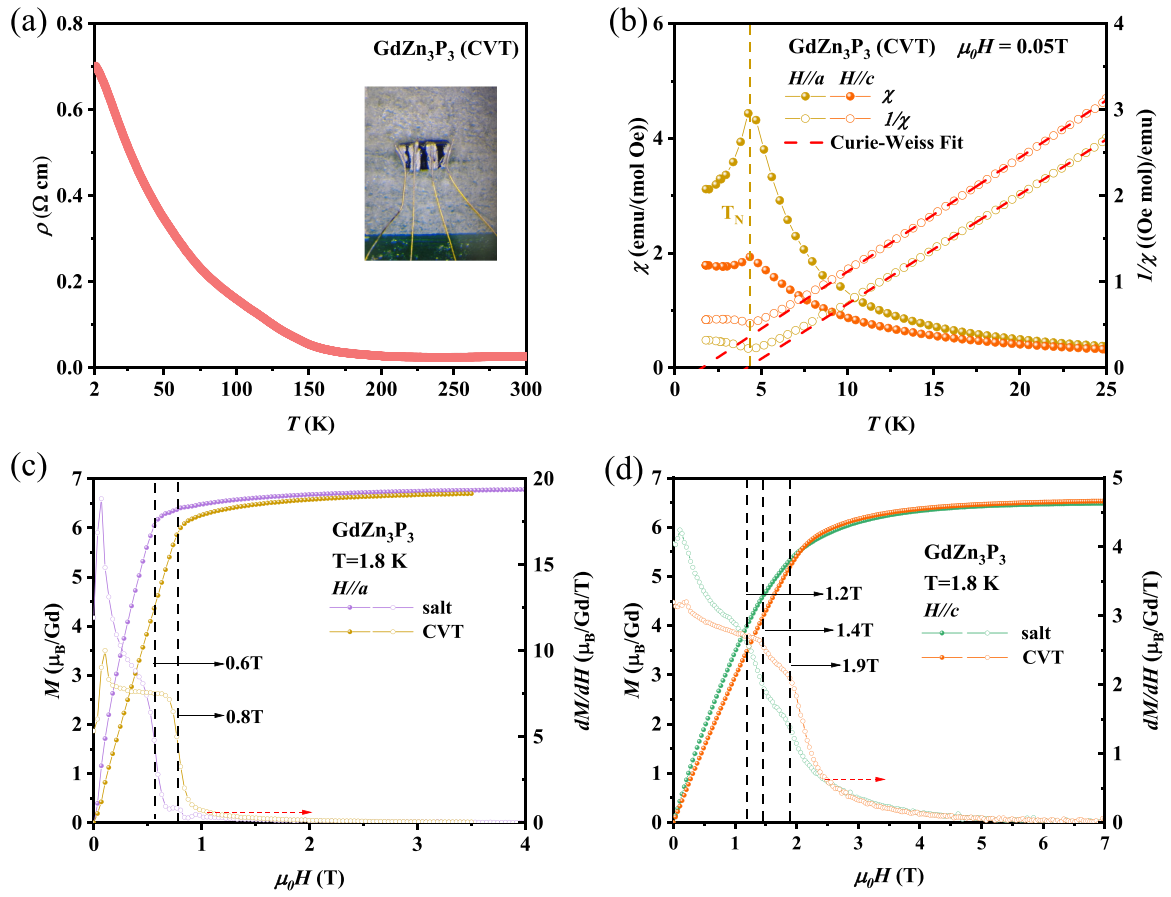


FIG. 9. (a) The temperature-dependent resistivity of GdZn<sub>3</sub>P<sub>3</sub> (CVT) single crystal between 2 K and 300 K. (b) The temperature-dependent magnetic susceptibility  $\chi(T)$  of GdZn<sub>3</sub>P<sub>3</sub> (CVT) single crystal between 1.8 and 25 K measured with magnetic field along  $a$  and  $c$  directions. Isothermal magnetization and the differential susceptibility  $dM/dH$  at 1.8 K with the magnetic field applied along the  $a$  (c) and  $c$  (d) directions for GdZn<sub>3</sub>P<sub>3</sub> (salt) and GdZn<sub>3</sub>P<sub>3</sub> (CVT) single crystals.

crystal below 3 K is absent in the CVT-grown sample (see Fig. S12).

The saturated magnetization reaches  $M_S \sim 6.7 \mu_B$  for  $H // a$  and  $M_S \sim 6.5 \mu_B$  for  $H // c$  at 1.8 K [Fig. 9(c) and 9(d)], consistent with the values observed in GdZn<sub>3</sub>P<sub>3</sub> (salt). From the corresponding  $dM/dH$  curves shown in Fig. 9(d), the metamagnetic transition for  $H // a$  occurs at approximately  $\mu_0 H_{ab} = 0.8 \text{ T}$ , which is higher than the in-plane transition field of  $\mu_0 H_{ab} = 0.6 \text{ T}$  observed in GdZn<sub>3</sub>P<sub>3</sub> (salt). For  $H // c$ , the metamagnetic transition field at  $\mu_0 H_{c1} = 1.4 \text{ T}$  is higher than the  $\mu_0 H_{c1} = 1.2 \text{ T}$  observed in GdZn<sub>3</sub>P<sub>3</sub> (salt), and both samples exhibit another metamagnetic transition near  $\mu_0 H_{c2} = 1.9 \text{ T}$ . Finally, the two curves are on top of each other above 2.5 T. In addition, small anomalous features have also been observed near zero field for both in-plane and out-of-plane orientations in Figs. 9(c) and 9(d), the microscopic origin of which requires further investigation.

The resistivity of the GdZn<sub>3</sub>P<sub>3</sub> (CVT) crystal at 300 K is measured to be  $27 \text{ m}\Omega \text{ cm}$ , which lies between the values observed for the polycrystalline sample ( $20 \text{ }\Omega \text{ cm}$  at 300 K) and the salt-grown crystal ( $4 \text{ m}\Omega \text{ cm}$  at 290 K). It is noteworthy that the magnitude of low-temperature  $\chi(T)$  for GdZn<sub>3</sub>P<sub>3</sub> (CVT) single crystal is basically comparable to that observed in GdZn<sub>3</sub>P<sub>3</sub> single crystals reported in Ref. [29]. Moreover, both crystals exhibit comparable resistivity values at 300 K

and are situated within the semiconductor-to-metal transition region. In contrast to the single crystals in Ref. [29], whose resistivity gradually decreases with decreasing temperature, the GdZn<sub>3</sub>P<sub>3</sub> (CVT) single crystal exhibits weak semiconducting behavior, as previously noted.

The metallic conductivity observed in the GdZn<sub>3</sub>P<sub>3</sub> (salt) single crystal is likely attributable to the relatively high carrier density introduced during crystal growth. Through the RKKY exchange interaction, the charge carriers may also influence the magnetic properties, such as enhancing the magnetic anisotropy in this  $S = 7/2$  Gd<sup>3+</sup>-containing system. Correspondingly, the anisotropy factor  $\chi_a(T)/\chi_c(T)$  at  $T_N$  increases from 2.3 in the semiconducting CVT-grown sample to 4.2 in the metallic salt-flux grown sample. Consequently, our investigation of the physical properties of both salt-flux-grown and CVT-grown GdZn<sub>3</sub>P<sub>3</sub> crystals, combined with a comparative analysis of the results from Ref. [29], clearly demonstrates that the A-type AFM ordering of Gd<sup>3+</sup> ions originates from the RKKY interaction mediated by hole-type carriers.

Such intriguing behavior has also been observed in topological materials EuCd<sub>2</sub>As<sub>2</sub>, where both metallic salt-flux-grown crystal and semiconducting Sn-flux-grown crystal exhibit A-type antiferromagnetic transitions at  $T_N \sim 9.3 \text{ K}$  [34,35,38]. Under  $H // c$ , the magnitude of  $\chi(T)$

of  $\text{EuCd}_2\text{As}_2(\text{Sn})$  ( $\sim 2.5$  emu/mol) is smaller than that of  $\text{EuCd}_2\text{As}_2(\text{salt})$  ( $\sim 4$  emu/mol), while for  $H//ab$ , the magnitude of  $\chi(T)$  of  $\text{EuCd}_2\text{As}_2(\text{Sn})$  ( $\sim 5$  emu/mol) is about half that of  $\text{EuCd}_2\text{As}_2(\text{salt})$  ( $\sim 12$  emu/mol) below  $T_N$ . According to Ref. [34], further Eu vacancies would drive the  $\text{EuCd}_2\text{As}_2$  system into metallic FM state below  $T_C = 26.4$  K.

#### IV. DISCUSSION AND CONCLUSION

For the magnetic topological system  $\text{EuMnSb}_2$  [71–73] and  $\text{EuCd}_2\text{As}_2$  [34], both materials exhibit narrow-gap semiconductor behavior. The presence of vacancies at Eu sites can introduce hole-type charge carriers and lead to metallic conductivity within the system. Concurrently, the enhanced RKKY interaction between the Eu magnetic moments and these charge carriers slightly alter the temperature-dependent magnetic ordering behavior, resulting in modifications to the spin configuration in the ground-state.

According to Ref. [49], the  $\text{RCd}_3\text{P}_3$  would be semiconductor with a band gap between 0.6–0.8 eV, verified by the DFT calculation and the polycrystal results. Given the narrow band gap of the  $\text{RCd}_3\text{P}_3$  family, these compounds are likely dopable into a metallic state as well. According to previous research, single crystals of  $\text{RCd}_3\text{X}_3$  ( $R = \text{La, Ce}$ ;  $X = \text{P, As}$ ) exhibit metallic behavior, with the resistivity on the order of 10 m $\Omega$  cm (1 m $\Omega$  cm) for P (As) compounds [47,48], yet the exact origin still unclear.

Here, the ZnP bond instability in  $\text{GdZn}_3\text{P}_3$  has been demonstrated through both x-ray diffraction and STEM measurement. Our atomic resolution STEM measurement reveals the presence of interstitial P atoms close to the ZnP honeycomb lattice, which would act as electron acceptors, thereby introducing hole-type carriers into the semiconducting system. Our investigation gives important insight into the metallic behavior of  $\text{RZn}_3\text{P}_3$  and  $\text{RCd}_3\text{X}_3$  ( $R = \text{La, Ce}$ ;  $X = \text{P, As}$ ) single crystals, thus it would be interesting to conduct the STEM measurement on the above single crystal to reveal its origin in atomic scale.

As mentioned above, the structural instability in ZnP layer could influence the rare-earth magnetic ion triangular layer through the charge carrier mediated RKKY magnetic interaction as observed in  $\text{GdZn}_3\text{P}_3$ ; meanwhile, the influence could also be realized through the CEF effect for rare-earth magnetic ion with orbital moment such as in  $\text{RZn}_3\text{P}_3$  ( $R = \text{Ce, Pr, Nd, and Sm}$ ) system [47,48,50], which deserves further investigation in the future. Given the relatively high AFM transition at  $T_N = 4.5$  K, our Hall measurements reveal a clear nonlinear AHE at low carrier density in metallic  $\text{GdZn}_3\text{P}_3$  crystal. In summary, these observations suggest that the  $\text{GdZn}_3\text{P}_3$  system offers a promising platform for investigating the interplay between charge carriers and magnetism in two-dimensional  $\text{RM}_3\text{X}_3$  compounds.

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#### DATA AVAILABILITY

The data that support the findings of this article are not publicly available upon publication because it is not technically feasible and/or the cost of preparing, depositing, and hosting the data would be prohibitive within the terms of this research project. The data are available from the authors upon reasonable request.

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