Tetragonal Cs$_{1.17}$In$_{0.81}$Cl$_3$: A Charge-Ordered Indium Halide Perovskite Derivative

Xiaoyan Tan, Peter W. Stephens, Mylene Hendrickx, Joke Hademann, Carlo U. Segre, Mark Croft, Chang-Jong Kang, Zheng Deng, Saul H. Lapidus, Sun Woo Kim, Changqing Jin, Gabriel Kotliar, Martha Greenblatt

Department of Chemistry and Chemical Biology and Department of Physics and Astronomy, Rutgers, The State University of New Jersey, Piscataway, New Jersey 08854, United States

Department of Physics and Astronomy, State University of New York, Stony Brook, New York 11794, United States

EMAT, University of Antwerp, Groenenborgerlaan 171, B-2020 Antwerp, Belgium

Department of Physics and CSRI, Illinois Institute of Technology, Chicago, Illinois 60616, United States

Institute of Physics, School of Physics, University of Chinese Academy of Sciences, Chinese Academy of Sciences, P.O. Box 603, Beijing 100190, P. R. China

Advanced Photon Source, Argonne National Laboratory, Argonne, Illinois 60439, United States

Department of Chemical Education, Chosun University, Gwangju 61452, South Korea

Supporting Information

ABSTRACT: Polycrystalline samples of Cs$_{1.17}$In$_{0.81}$Cl$_3$ were prepared by annealing a mixture of CsCl, InCl, and InCl$_3$, stoichiometric for the targeted CsInCl$_3$. Synchrotron powder X-ray diffraction refinement and chemical analysis by energy dispersive X-ray indicated that Cs$_{1.17}$In$_{0.81}$Cl$_3$ is a tetragonal distorted perovskite derivative ($I4/m$), which is the thermodynamically stable product. The refined unit cell parameters and space group were confirmed by electron diffraction. In the tetragonal structure, In$^+$ and In$^{3+}$ are located in four different crystallographic sites, consistent with their corresponding bond lengths. In1, In2, and In3 are octahedrally coordinated, whereas In4 is at the center of a pentagonal bipyramid of Cl because of the noncooperative octahedral tilting of In4Cl$_6$. The charged-ordered In$^+$ and In$^{3+}$ were also confirmed by X-ray absorption and Raman spectroscopy. Cs$_{1.17}$In$_{0.81}$Cl$_3$ is the first example of an inorganic halide double perovskite derivative with charged-ordered In$^+$ and In$^{3+}$. Band structure and optical conductivity calculations were carried out with both generalized gradient approximation (GGA) and modified Becke–Johnson (mBJ) approach; the GGA calculations estimated the band gap and optical band gap to be 2.27 eV and 2.4 eV, respectively. The large and indirect band gap suggests that Cs$_{1.17}$In$_{0.81}$Cl$_3$ is not a good candidate for photovoltaic application.

INTRODUCTION

Recently, inorganic halide perovskites and related compounds have gained attention because of their promising application in optoelectronic devices such as photodetectors, solar cells, light-emitting diodes, semiconductor lasers, and so forth. In particular, cesium-based halide perovskites with formula CsM$_6$X$_3$ (M = Pb, Sn; X = Cl, Br, I) have shown exceptional optoelectronic properties. CsPbI$_3$ quantum dots have been incorporated into solar cells, which display a power conversion efficiency of 10.77%. The efficiency can be further increased to 11.33% in CsPb$_{0.9}$Sn$_{0.1}$I$_2$ by substituting Pb with Sn and I with Br. The highest efficiency of solar cells based on CsPbI$_3$ quantum dot films reaches 13.45%, which is the highest among all quantum-dot-based solar cells. CsPbBr$_3$-based green light-emitting diodes exhibit a high electroluminescence quantum efficiency of 10.4%, which are the highest in all perovskite green light-emitting diodes.

CsPbX$_3$ has outstanding properties because of the toxicity of Pb, both experimental and theoretical scientists are searching for alternative inorganic halide perovskites. Filip et al. used computational screening methods and explored the possible cesium metal halide perovskites with other divalent metal ions such as Mg$^{2+}$, V$^{2+}$, Mn$^{2+}$, Ni$^{2+}$, Cd$^{2+}$, Hg$^{2+}$, Ga$^{3+}$, and In$^{2+}$. The proposed CsInCl$_3$ and CsInBr$_3$ compounds are of interest because of the unusual 2+ valence state of In, which has not been reported so far. Later, Körlb et al. also carried...
out extensive density functional theory (DFT) calculations to explore novel inorganic perovskites and identified CsInCl₃ and RbInBr₃ as small gap semiconductors. More recent work by Kang and Kotliar studied systematically the structures of AInX₃ compounds (A = alkali metals, X = F or Cl) and identified the low-lying structures of these materials and showed that CsInCl₃ should be thermodynamically stable and thus should form, providing further motivation for our studies. Recent experimental study of the similar compound CsTlX₃ (X = F, Cl) found that Tl disproportionated into Tl⁺ and Tl³⁺. Therefore, we expected that In might disproportionate into In⁺ and In³⁺ in CsInCl₃ as well. To explore the existence of CsInCl₃ compound, its crystal structure, and the oxidation states of In, we embarked on the experimental work to synthesize and investigate the properties of CsInCl₃.

Another motivation for studying CsInCl₃ arises from its possible superconductivity, observed in related compounds. An example is the well-known perovskite BaBiO₃, an insulator with mixed valence Bi⁺ and Bi⁵⁺, which becomes superconducting by hole doping in Ba₁₋ₓKₓBiO₃ and BaBi₁₋ₓPbₓO₃ because of the strongly phonon-coupled bands across the Fermi level. CsTlX₃ has the same valence electron count as BaBiO₃, which is expected to show a similar band structure. Theoretical calculations predict the superconducting critical temperatures of doped CsTlF₃ and CsTlCl₃ to be ~30 and ~20 K, respectively. Although experimental work on CsTlX₃ has not yet discovered the superconductivity in these phases so far, it is of interest to explore analogous CsInCl₃ for future discovery.

In this manuscript, we report that polycrystalline Cs₁₁₂In₉₈Cl₃ with the space group I₄/m. (a) The xy layer at z = 0, (b) the xy layer at z = 1/4, and (c) a perspective view of the structure viewed along the y axis. Color code: In₁ = magenta, In₂ = red, In₃ = green, In₄ = orange, Cs = grey, and Cl = black.

EXPERIMENTAL METHODS

Starting Materials and Synthesis. The nominal CsInCl₃ phase was prepared by conventional solid-state methods, similar to ones used for tetragonal CsTlCl₃ (I₄/m), with anhydrous CsCl (wt 99.99%, Alfa Aesar), InCl (wt 99.999%, Alfa Aesar), and InCl₃ (wt 99.999%, Alfa Aesar). The starting materials were mixed in the ratio of CsCl:InCl:InCl₃ = 2:1:1, ground thoroughly, and loaded into a silica tube, which was sealed under vacuum (<10⁻² mbar). The ampule was ramped up to 923 K within 24 h, kept at this temperature for 24 h, and then cooled down to room temperature over a course of 24 h. All sample preparations were carried out in an Ar-filled glove box with O₂ and H₂O concentration below 1 ppm.

Laboratory and Synchrotron Powder X-ray Diffraction (SPXD). Samples were measured in the 2θ range from 10 to 80° and collected for 1 h on a Bruker D8 Advance diffractometer (Cu Kα, λ = 1.5418 Å) with a SOL-X solid state detector. Room temperature SPXD (λ = 0.412723 Å) was collected in the 2θ range from 0.5 to 50° at the 11-BM-B beam line at the Advanced Photon Source (APS), Argonne National Laboratory. Diffraction data analysis and Rietveld refinement were performed with TOPAS-Academic.

Chemical Analysis. The chemical composition of the samples was determined by energy dispersive X-ray (EDX) analysis with a JEOL JSM-7500F scanning electron microscope (SEM) equipped with an INCA Energy 250 EDS system (Oxford Instruments). Sample preparation was carried out in the glove box to avoid contact with air, and the Cs L, In L, and Cl K lines were used for the composition quantification.

A separate chemical composition analysis was performed on a FEI Osiris transmission electron microscope (TEM), operated at 200 kV equipped with a Super-X detector. A vacuum transfer holder was used to avoid contact with air, and the Cs L, In L, and Cl K lines were used for the composition quantification.

Electron Diffraction. The samples for the TEM analysis were prepared in the glove box by crushing the powder of the sample and gently depositing the dry powder on a copper grid covered with a holey carbon film. The sample was transported and inserted into the TEM without contact with air with a vacuum transfer holder. Selected area electron diffraction (SAED) patterns were performed with a FEI Tecnai G2 TEM.

Thermogravimetric Analysis (TGA) and Differential Scanning Calorimetry (DSC). A powder sample of about 25 mg was loaded into an alumina crucible in a SERT Q600 TA instrument and measured between room temperature and 1000 °C at a heating rate of 10 °C/min under argon flow.

X-ray Absorption Spectroscopy (XAS). XAS data were collected at the indium K-edge in fluorescence at the Materials Research Collaborative Access Team (MRCAT) insertion device beam line (Sector 10-ID) on a powder-filled Kapton capillary at room temperature. Data were normalized and background subtracted with Athena and IFEFFIT. White line intensities were obtained with Fityk with a Gaussian peak function plus an arctangent step (constrained to the peak energy).

Raman Spectroscopy. Micro Raman measurements have been carried out with an inVia Raman microscope (Renishaw) and excited with a solid-state laser (λexc = 532 nm).

DFT Calculations. The all-electron full-potential linearized augmented plane-wave (FLAPW) method implemented in WIEN2k was adopted to calculate the electronic structure. A generalized gradient approximation (GGA) functional was chosen to calculate the electronic structure. Because GGA underestimates a band gap, we used the modified Becke–Johnson (mBJ) exchange
RESULTS AND DISCUSSION

Chemical Analysis and Crystal Structure. A polycrystalline light yellow product was obtained, and powder X-ray diffraction (PXRD) analysis confirmed the major phase to be “CsInCl₃”, with ~5% of the Cs₅In₇Cl₂H₂O impurity. The sample is hygroscopic, and the color of sample becomes darker when it is exposed to air. Numerous efforts to synthesize CsInF₃ and RbInCl₃ with similar approach at different temperatures failed to yield the targeted phases.

SEM-EDX and TEM-EDX measurements were carried out to investigate the chemical composition of the “CsInCl₃” product. Both EDX studies clearly confirm the presence of the Cs, In, and Cl elements (representative spectrum in Figure S1). The TEM-EDX study of 20 different crystals reveals a molar ratio of Cs:In:Cl = 0.26(2):0.18(1):0.56(2), which is close to a target of 1:1:3 but with an excess of Cs and less In. This is equivalent to a Cs/In ratio of 1.44 ± 0.14. These results are also confirmed by a statistical SEM-EDX study taken from 50 different crystals, which shows a molar ratio of Cs:In:Cl = 0.23(2):0.16(1):0.60(2) with a Cs/In ratio equal to 1.44 ± 0.16. Thus, the chemical analysis indicates that the composition of the material produced is a Cs rich, In deficient phase of formula Cs₁.₁₇In₀.₈₁Cl₃. We speculate that the difference from stoichiometry is not an intrinsic property of the nominal CsInCl₃ phase but due to unknown side reaction(s).

SPXD of the Cs₁.₁₇In₀.₈₁Cl₃ sample is very similar to that of CsTICl₃, and indeed, initial Rietveld refinement results indicate that it is isostructural with tetragonal CsTICl₃. In this structure, shown in Figure 1, there are two Cs sites, four In sites, and seven Cl sites. The SEM-EDX results imply that structure, shown in Figure 1, there are two Cs sites, four In sites, and seven Cl sites. The SEM-EDX results imply that structure, shown in Figure 1, there are two Cs sites, four In sites, and seven Cl sites. The SEM-EDX results imply that structure, shown in Figure 1, there are two Cs sites, four In sites, and seven Cl sites.

Two of the nominal In sites, In1 and In3, are octahedrally coordinated, with In–Cl bond distances between 2.50–2.64 Å, close to the value expected for the In³⁺–Cl⁻ distances based on Shannon sum of radii (2.61 Å). The calculated bond valence values for the sum of In1 and In3 are 3.06 and 3.14, respectively. Therefore, the charge of In on In1 and In3 sites can be assigned to 3+. The observed In⁵⁺–Cl⁻ distances are also consistent with distances in Cs₅In₇Cl₂ (2.41–2.65 Å) and Cs₅InCl₇H₂O (2.47–2.71 Å).

The remaining two nominal In sites have significantly longer In–Cl bond distances (3.00–3.55 Å), indicating a lower oxidation state on In2 and In4 sites. On the basis of the similar bond distances between Cs₁.₁₇In₀.₈₁Cl₃ and CsTICl₃ (Table 2), one could conclude that In⁴⁺ occupies In2 and In4 sites, similar to the case of Tl⁺ on Tl2 and Tl4 sites. Although the Shannon radius of In⁺ is not assigned, the estimated In⁺–Cl⁻ distance should be close to the expected Tl⁺–Cl⁻ distance (3.31 Å) for a six-coordinated environment, which agrees with In⁺–Cl⁻ distances in Cs₁.₁₇In₀.₈₁Cl₃. The observed In⁺–Cl⁻ distances are also comparable with those in red InCl (2.87–3.26 Å) and yellow InCl (2.90–3.53 Å). In2 is octahedrally coordinated, whereas In⁴⁺ is at the center of a pentagonal bipyramid of Cl. It is therefore not surprising that In⁴⁺–Cl⁻ distances (3.05–3.55 Å) are longer than In²⁺–Cl⁻ distances (3.00–3.02 Å). The long In⁴⁺–Cl⁻ distances are also observed in the mixed valence compound, In₃Cl₆, where the average In⁻–Cl distance is 3.58 Å. There, the In⁴⁺/In³⁺ charge order, no In²⁺ in Cs₁.₁₇In₀.₈₁Cl₃ similar to that of Tl⁺/Tl³⁺ in CsTlCl₆₅₁₃₄.,

We now turn to the occupancy of the metal sites. If only In is placed in the In sites, refined occupancies in three of the sites are essentially unity and 0.93 ± 0.01 in site In2 for a sample composition of Cs₁.₁₇In₀.₈₁Cl₃. This refinement is presented as model I in Table S1. However, the EDX results imply that there must be a significant amount of Cs in the nominal In sites as discussed below. Isotropic thermal parameters were refined independently for each site, except for those of the Cl sites, which had a common thermal factor. The Rietveld fit is shown in Figure 2, and the refined crystallographic data are summarized in Table 1. Details of the refinement, including raw data and computed fit, have been deposited as a cif file in the inorganic crystal structure database with CCDC number 1877027.

Figure 2. Rietveld refinement of synchrotron powder X-ray diffraction pattern of Cs₁.₁₇In₀.₈₁Cl₃, including ~5% of the Cs₅In₇Cl₂H₂O impurity. Black dots are the observed pattern, the red line is the calculated pattern, vertical black bars are the Bragg positions of Cs₁.₁₇In₀.₈₁Cl₃ (upper row) and Cs₅In₇Cl₂H₂O impurity (lower row), and the green line is the difference between observed and calculated patterns. The most visible peaks of the impurity phase are circled in blue.

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Table 1. Refined Crystallographic Data of Cs$_{1.17}$In$_{0.81}$Cl$_3$ According to Model II Described in the Text

<table>
<thead>
<tr>
<th>compound</th>
<th>Cs$<em>{1.17}$In$</em>{0.81}$Cl$_3$</th>
<th>Cs$<em>{1.17}$In$</em>{0.81}$Cl$_3$, 354.20</th>
</tr>
</thead>
<tbody>
<tr>
<td>X-ray wavelength</td>
<td>$\lambda = 0.412723 \text{ Å}$</td>
<td>$a = b = 17.1098(1) \text{ Å}$, $c = 11.0523(1) \text{ Å}$, $a = \beta = \gamma = 90^\circ$; 3235.50(5) Å$^3$</td>
</tr>
<tr>
<td>space group</td>
<td>$I4/m$, #87</td>
<td></td>
</tr>
<tr>
<td>Rietveld criteria of fit for powder diffraction refinements$^a$</td>
<td>$R_p = 7.43%$, $R_w = 9.85%$, $R_{wp} = 5.21%$</td>
<td></td>
</tr>
<tr>
<td>parameters refined</td>
<td>62</td>
<td></td>
</tr>
<tr>
<td>parameter restraints$^a$</td>
<td>9</td>
<td></td>
</tr>
</tbody>
</table>

<table>
<thead>
<tr>
<th>nominal atom, (valence charge)</th>
<th>site multiplicity and Wyckoff symbol</th>
<th>$x$</th>
<th>$y$</th>
<th>$z$</th>
<th>fractional occupancy model II</th>
<th>$B_{iso}$</th>
</tr>
</thead>
<tbody>
<tr>
<td>Cs1 (+1)</td>
<td>4d</td>
<td>0</td>
<td>$\frac{1}{2}$</td>
<td>$\frac{1}{4}$</td>
<td>1</td>
<td>5.7(2)</td>
</tr>
<tr>
<td>Cs2 (+1)</td>
<td>16i</td>
<td>0.2108(1)</td>
<td>0.1091(1)</td>
<td>0.2773(1)</td>
<td>1</td>
<td>4.64(5)</td>
</tr>
<tr>
<td>In1 (+3)</td>
<td>2a</td>
<td>0</td>
<td>0</td>
<td>0</td>
<td>In: 1</td>
<td>1.41(1)</td>
</tr>
<tr>
<td>In2 (+1)</td>
<td>2b</td>
<td>0.1753(1)</td>
<td>0.1753(1)</td>
<td>0</td>
<td>In: 0.629(3) Cs: 0.371(3)</td>
<td>5.1(4)</td>
</tr>
<tr>
<td>In3 (+3)</td>
<td>8h</td>
<td>0.0972(4)</td>
<td>0.3073(4)</td>
<td>0.2688(4)</td>
<td>1</td>
<td>3.27(7)</td>
</tr>
<tr>
<td>In4 (+1)</td>
<td>8h</td>
<td>0.2616(6)</td>
<td>0.2470(6)</td>
<td>0</td>
<td>1</td>
<td>3.27(7)</td>
</tr>
<tr>
<td>Cl1 (−1)</td>
<td>4e</td>
<td>0</td>
<td>0</td>
<td>0</td>
<td>0.227(1)</td>
<td>1</td>
</tr>
<tr>
<td>Cl2 (−1)</td>
<td>16i</td>
<td>0.1372(5)</td>
<td>0.0586(6)</td>
<td>0</td>
<td>1</td>
<td>3.27(7)</td>
</tr>
<tr>
<td>Cl3 (−1)</td>
<td>8h</td>
<td>0.3771(6)</td>
<td>0.0524(7)</td>
<td>0</td>
<td>1</td>
<td>3.27(7)</td>
</tr>
<tr>
<td>Cl4 (−1)</td>
<td>8h</td>
<td>0.1529(7)</td>
<td>0.4539(7)</td>
<td>0</td>
<td>1</td>
<td>3.27(7)</td>
</tr>
<tr>
<td>Cl5 (−1)</td>
<td>8h</td>
<td>0.4579(8)</td>
<td>0.3297(6)</td>
<td>0</td>
<td>1</td>
<td>3.27(7)</td>
</tr>
</tbody>
</table>

$^a$R$_{wp} = \sqrt{\frac{\sum w_i(y_i^\text{calc} - y_i^\text{obs})^2}{\sum w_i(y_i^\text{calc} - y_i^\text{calc})^2}}$, $R_p = \sqrt{\frac{\sum w_i(y_i^\text{calc} - y_i^\text{obs})^2}{\sum w_i(y_i^\text{calc})^2}}$, $\chi^2 = (R_p/R_{wp})^2$, where $y_i^\text{calc}$ and $y_i^\text{obs}$ are the calculated and observed intensities at the ith point in the profile, the weight $w_i$ is $1/\sigma^2$ from counting statistics, with the same normalization factor, N is the number of points in the measured profile, and P is the number of refined parameters. $R_{wp}$ is the expected value of $R_p$ if the only deviation of the data from the model is due to statistical errors.$^b$All Cl thermal parameters were restrained to be equal plus three restraints on In:Cs vacancy composition of In$_2$ and In$_4$ sites.

Table 2. Selected Bond Distances in Cs$_{1.17}$In$_{0.81}$Cl$_3$ and Tl–Cl Bond Distances in CsTlCl$_3$\textsuperscript{17a}

<table>
<thead>
<tr>
<th>bond</th>
<th>Cs–Cl distances (Å)</th>
<th>bond</th>
<th>In–Cl distances (Å)</th>
<th>bond</th>
<th>Tl–Cl\textsuperscript{17} distances (Å)</th>
</tr>
</thead>
<tbody>
<tr>
<td>Cs1–Cl4</td>
<td>3.59(1) × 4</td>
<td>In1–Cl1</td>
<td>2.50(1) × 2</td>
<td>Tl1–Cl3</td>
<td>2.45(3) × 4</td>
</tr>
<tr>
<td>Cs1–Cl2</td>
<td>3.66(1) × 4</td>
<td>In1–Cl3</td>
<td>2.55(1) × 4</td>
<td>Tl1–Cl2</td>
<td>2.56(3) × 2</td>
</tr>
<tr>
<td>Cs1–Cl6</td>
<td>3.89(1) × 4</td>
<td>In2–Cl7</td>
<td>3.00(1) × 4</td>
<td>Tl2–Cl1</td>
<td>2.99(3) × 2</td>
</tr>
<tr>
<td>Cs2–Cl3</td>
<td>3.42(1)</td>
<td>In2–Cl1</td>
<td>3.02(1) × 2</td>
<td>Tl2–Cl7</td>
<td>3.07(3) × 4</td>
</tr>
<tr>
<td>Cs2–Cl5</td>
<td>3.51(1)</td>
<td>In3–Cl7</td>
<td>2.44(1)</td>
<td>Tl3–Cl3</td>
<td>2.39(3)</td>
</tr>
<tr>
<td>Cs2–Cl6</td>
<td>3.56(1)</td>
<td>In3–Cl6</td>
<td>2.49(1)</td>
<td>Tl3–Cl2</td>
<td>2.62(1) × 2</td>
</tr>
<tr>
<td>Cs2–Cl7</td>
<td>3.64(1)</td>
<td>In3–Cl4</td>
<td>2.52(1)</td>
<td>Tl3–Cl4</td>
<td>2.69(3)</td>
</tr>
<tr>
<td>Cs2–Cl2</td>
<td>3.95(1)</td>
<td>In3–Cl2</td>
<td>2.56(1) × 2</td>
<td>Tl3–Cl6</td>
<td>2.71(3)</td>
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<tr>
<td>Cs2–Cl1</td>
<td>3.83(1)</td>
<td>In3–Cl5</td>
<td>2.64(1)</td>
<td>Tl3–Cl5</td>
<td>2.84(3)</td>
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<tr>
<td>Cs2–Cl7</td>
<td>3.93(1)</td>
<td>In4–Cl5</td>
<td>3.05(1)</td>
<td>Tl4–Cl5</td>
<td>2.82(3)</td>
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<tr>
<td>Cs2–Cl2</td>
<td>3.95(1)</td>
<td>In4–Cl4</td>
<td>3.08(1)</td>
<td>Tl4–Cl4</td>
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<tr>
<td>Cs2–Cl5</td>
<td>3.97(1)</td>
<td>In4–Cl2</td>
<td>3.09(1) × 2</td>
<td>Tl4–Cl2</td>
<td>3.01(1) × 2</td>
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<tr>
<td>Cs2–Cl1</td>
<td>4.10(1)</td>
<td>In4–Cl3</td>
<td>3.20(1)</td>
<td>Tl4–Cl3</td>
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<tr>
<td>Cs2–Cl4</td>
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<td>In4–Cl6</td>
<td>3.54(1)</td>
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<tr>
<td>Cs2–Cl3</td>
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<td>In4–Cl3</td>
<td>3.55(1)</td>
<td>Tl4–Cl3</td>
<td>3.55(3)</td>
</tr>
</tbody>
</table>

$^a$For each metal atom, bonds are listed in order of increasing length.

valence charge imbalance. The most plausible solution is that some fraction of In in those sites must be In$^{3+}$ instead of In$^{1+}$. In the tetragonal ab plane, In$_2$C$_{16}$ octahedra corner share by four In$_4$C$_7$ octahedra in fourfold axis symmetry, whereas In$_1$C$_6$ octahedra edge-share with In$_4$C$_7$ pentagonal bipyramids in the same fashion. Then, these two symmetric segments are connected by corner-sharing In$_3$C$_6$ octahedra and In$_4$C$_7$ pentagonal bipyramids. Layers centered with In$_2$C$_6$ and In$_1$C$_6$ octahedra are shown in Figure 1a,b, respectively. These two layers alternate every half a unit cell along the tetragonal c axis with a corner-sharing connection (Figure 1c). The fully occupied A-site Cs atoms are located in the space between these InCl$_{16}$, In$_2$C$_{16}$, In$_3$C$_{16}$, and In$_4$C$_7$ polyhedra.

One interesting feature of Cs$_{1.17}$In$_{0.81}$Cl$_3$ is that In$^{4+}$ is seven-coordinated with Cl$^-$ and forms an In$_4$Cl$_7$ pentagonal bipyramid. The irregular coordination environment and the long bond distances for In$^{4+}$ may be attributed to the almost doubly filled In 5s atomic orbital as shown in the study of binary indium bromides.$^{15–17}$ Interestingly, the lone pair 5s$^2$
electrons of In2+ are not stereoactive, whereas those of In4+ appear to be active; a lone pair distortion can be switched on or off by the environment and does not always distort in its vicinity. If we disregard for the moment the nonstoichiometric composition and presence of Cs in the In sites, nominal “CsIn4+In3+Cl4−” could be understood as “Cs5In5Cl16−”, a modified form of a double perovskite halide with a B site charge order in the formula of A2BB′X6. In the double perovskite, the three-dimensional (3D) framework is formed by corner-sharing BX6 and B′X6 octahedra, with A site atoms occupying the cubo-octahedral void of the 3D structure. The modified double perovskite tetragonal structure (I4/m) of Cs1.17In0.81Cl3 stems from the ~45° rotation in the ab plane of one-fifth of the B′X6 octahedra (In1Cl6) in the double perovskite structure. This In1Cl6 rotation results in the change of an adjacent In4 site coordination environment from In4Cl6 octahedra to the In4Cl4 pentagonal bipyramid, with long In4−Cl bond distances. Moreover, edge-sharing is created between In1Cl6 and In4Cl4. The interruption of connected corner-sharing caused by the B′X6 octahedra rotation is defined as a noncooperative octahedral tilting (NCOT) in double perovskites. Other examples in halides with a tetragonal structure (I4/m) include β-K3AlF6, low temperature Rb2KCrF6 and β-Rb2KGaF6, in which the K ions on the B site are also coordinated with seven F atoms and form pentagonal bipyramids. NCOT also has been reported in oxides and oxyfluorides. In the modified double perovskite structure of Cs1.17In0.81Cl3, the environment of A site ions is also change, where Cs1 still remains in the cubo-octahedral geometry, but Cs2 connects with eleven Cl and forms an irregular polyhedron (Figure 3).

Electron Diffraction. To confirm the refined crystal structure based on SPXD data, selected area electron diffraction (SAED) patterns of the Cs1.17In0.81Cl3 sample are taken from different zone axes ([001], [001], [210], [113]/[135], [353], and [425]) (Figure 4). All SAED patterns could be indexed with the tetragonal cell parameters: a = b ≈ 17.11 Å, c ≈ 11.052 Å, and α, β, γ = 90° and corresponded to the following reflection conditions: hkl: h + k + l = 2n; hk0: h + k = 2n; 0kl: k + l = 2n; 00k: k = 2n.

To check the reflection condition 00l (whether it is l = 2n or l = 4n), two tilt series of SAED patterns were obtained along the c-axis, shown in Figures S3 and S4. The intensity of the reflections 00l: l = 2n stays strong during the whole tilt series, indicating that they are not caused by double diffraction and thus do not corresponded to 00l: l = 4n. According to the reflection conditions and cell parameters, the possible space groups are I4, I4, I4/m, I422, I4mm, I42m, I4m2, or I4/mmm.

On the basis of very clear intensity distributions in the reflections, all space groups with Laue class 4/mmm can be excluded. Here, we use the magnified experimental SAED pattern of the zone axis [001] for justification (Figure 5). In Figure 5, the projected positions of planes perpendicular to the <a> and <110> directions and twofold axes along <a> and <110> are drawn. If these are mirror planes or twofold axes for the structure, the reflections connected by these symmetry elements should have equivalent intensities. It is clear that the intensity distribution of the reflections does not satisfy any of these symmetry elements. Therefore, this disagreement rules out all space groups with Laue class 4/mmm, that is, I422, I4mm, I42m, I4m2, or I4/mmm.

Sample Stability. The stability of the Cs1.17In0.81Cl3 phase was determined by thermogravimetric analysis (TGA) and differential scanning calorimetry (DSC) between 300 and 1273 K under Ar flow. Cs1.17In0.81Cl3 is thermally stable below 600 K without any phase transition or melting under the inert environment (Figure 6); however, the sample mass drops as the temperature increases from 600 to 1273 K. The mass drop between 600 and 900 K is probably due to sample melting. After 900 K, an endothermal peak shown in the DSC curve at 910 K indicates possible decomposition. The TGA–DSC curves are similar to those of tetragonal Cs2TlCl3, which is thermally stable up to 623–673 K, the temperature where that phase also starts melting.

Figure 4. SAED patterns of the Cs1.17In0.81Cl3 sample along the zone axes [100], [001], [210], [113]/[135], [353], and [425].

Figure 5. Magnified SAED pattern of zone axis [001]. The mirror plane perpendicular to x is indicated by “mx” and the twofold axis along x by an elliptical shape with index “x”.

Figure 3. Coordination environment for (a) Cs1 and (b) Cs2 in Cs1.17In0.81Cl3.
X-ray Absorption Spectroscopy (XAS). As a p-block element, the frontier bonding orbitals of In are the 5p/5s states with the configurations: In0 s2p1, In1+ s2p0 (reflecting the inert pair effect), and In3+ s0p0. X-ray absorption edges with final states in the p valence orbitals can be used to probe the valence/configuration of such p block elements. In Figure 7a, the In K near edge spectrum of Cs1.17In0.81Cl3 is shown along with the standards: elemental In0, In1+, InCl; and In3+, InCl3, and In2O3.

The In K near edge spectra are dominated by a strong “white line” (WL) peak feature because of 2s core transitions into empty 5p states. This WL feature rides upon a step feature because of the onset of transitions into the continuum. Typically, two points regarding the In valence/WL feature coupling should be noted: first, the increase in WL intensity with increasing valence and second, the chemical shift of the WL to higher energy is not always consistent with increasing valence.46−48 The chemical shifts in these materials appears to be too structure/ligand dependent for reliable interpretation, so the WL area will be focused upon.

The increased WL intensity, with increasing valence, is due partially to the increase of available empty p final states (e.g., between the In0 and In1+ states) and partially due to the increased transition matrix element via the increased localization of the p final states upon loss of screening with increasing valence (e.g., the loss of the s2 screening between the In1+ and In3+ states). The systematic WL intensity increase between In0, InCl, and InCl3 is clear. The WL intensity in InCl3 is also enhanced by an amount somewhat more than in InCl, indicating bonding/ligand dependence. It is important to note that the WL intensity of Cs1.17In0.81Cl3 is intermediate between the In1+ and In3+ standards (the sizable ligand/structure dependence in the In3+ standards is qualification to this statement). The plot of the Gaussian fitted WL areas versus In formal valence for the standard compounds (referred to above) is shown in Figure 8 along with a linear fit to the standard area data. Within the nonnegligible uncertainties in these estimates, these results support the average valence in Cs1.17In0.81Cl3 lying in the vicinity of In2+.

As another check on the charge separation on the sites in Cs1.17In0.81Cl3, FEFF 8.4 model45 calculations for all of the sites were performed with the near-edge results being shown in Figure 7b. In view of the substantial uncertainty in absolute energy in such modeling, the results have been presented relative to the calculated all-site-average peak for Cs1.17In0.81Cl3. The determined structure was used in this modeling. The model calculations indicate: the In2 and In4 sites have a smaller chemical shift and smaller WL area consistent with a smaller In valence and the In1 and In3 sites have a larger chemical shift and larger WL area consistent with...
a larger In valence. Thus, the model results are consistent with the experimental observations. It is worth noting that charge separation in solids can often be smaller than the stable full integral valence differences.

**Raman Spectroscopy.** To further compare Cs$_{1.17}$In$_{0.81}$Cl$_3$ with CsTlCl$_3$ and CsTl$_{1-x}$Hg$_x$Cl$_3$ ($0 \leq x < 1$), the Raman spectrum of Cs$_{1.17}$In$_{0.81}$Cl$_3$ was collected at room temperature under ambient pressure (Figure 9). For cubic CsTlCl$_3$, the calculated phonon frequency of the Tl−Cl stretching mode is 277 cm$^{-1}$. The experimental Raman spectra of both cubic (Fm$ar{3}$m) and tetragonal CsTlCl$_3$ (I4/m) show a major phonon frequency at 270 cm$^{-1}$, in agreement with the theoretical calculation results. In CsTl$_{1-x}$Hg$_x$Cl$_3$, active Raman Tl$^+$−Cl−Tl$^{3+}$ modes are also observed around 270 cm$^{-1}$. Here, the major phonon frequency of Cs$_{1.17}$In$_{0.81}$Cl$_3$ is at 278 cm$^{-1}$, which is close to values of Raman for Tl$^+$−Cl−Tl$^{3+}$ modes in CsTlCl$_3$ and CsTl$_{1-x}$Hg$_x$Cl$_3$. On the basis of the similarity observed in the structure and Raman spectra between Cs$_{8.17}$In$_{0.81}$Cl$_3$, CsTlCl$_3$, and CsTl$_{1-x}$Hg$_x$Cl$_3$, the active Raman mode observed in Cs$_{8.17}$In$_{0.81}$Cl$_3$ can be ascribed to In$^+$−Cl−In$^{3+}$.

**DFT Calculations.** To gain insight of the band structure and estimate the band gap, first-principles DFT calculations were carried out with both GGA (PBE version) and mBJ approaches (Figure 10). The calculations were conducted on the CsInCl$_3$ version of the structural model I, that is, without considering the apparent presence of Cs in the In sites or attendant vacancies in the crystal structure of Cs$_{8.17}$In$_{0.81}$Cl$_3$. The band structure with GGA approaches the lowest energy of the conduction band located at the N point, whereas the highest energy of valence band is located at the X point, which indicates an indirect band gap of 2.27 eV (Figure 10a).

As shown in Figure 10, the valence band density of states (DOS) around the Fermi level is mostly contributed by the In states, but the strong peaks located at lower energy are dominated by Cl states (Figure 10a,b,d,e). The In2 (1+, 5s$^2$) and In4 (1+, 5s$^3$) 5s bands are located in valence bands, whereas In1 (3+, 5s$^6$) and In3 (3+, 5s$^3$) 5s bands are located in conduction bands, which clearly support the two oxidation states in the structure as revealed in the structure and XANES analysis. A broader computational survey of compounds AlnX$_3$ (A = alkali metals, X = F or Cl) predicts that the most stable state of CsInCl$_3$, In has a single In$^{3+}$ site, but that it is only slightly more likely to occur than the phase described in this work (Tables S2 and S3).

GGA calculated corresponding optical conductivity predicts an optical gap of 2.42 eV (Figure 10c). In comparison with the GGA results, the calculated band structure using mBJ shows similar indirect band gap features but with a larger band gap of 2.83 eV (Figure 10d−f). In the valence bands, these strong peaks are shifted to lower energy. Consequently, the calculated optical gap using the mBJ approach is increased to 3.03 eV. The calculated band gap here is much larger than the literature report (<1 eV) of theoretical CsInCl$_3$. Because our calculations are carried out based on the experimental crystal structure, the estimated band gap shown here is more reliable than the reported values. The total DOS and estimated optical band gap of CsInCl$_3$ are similar to those of CsTlCl$_3$, which is predicted to exhibit superconductivity with proper doping. Therefore, it is worth carrying out further calculations and corresponding experiments on CsInCl$_3$ to explore the possibility of making it superconducting by design. It seems especially promising to seek the doping of this material, in view of its apparent natural tendency to prefer off stoichiometry. However, determining and controlling the charge state in sample preparation may be a significant challenge.

**CONCLUSIONS**

Although CsInCl$_3$ was originally predicted to be a perovskite, synchrotron X-ray diffraction data and X-ray fluorescence measurements show that the stable material has some Cs in the nominal In sites and adopts a tetragonal perovskite-related phase (I4/m) of composition Cs$_{8.17}$In$_{0.81}$Cl$_3$, with both corner-sharing and edge-sharing InCl$_6$ octahedra and InCl$_7$ pentagonal bipyramids. Cs$_{8.17}$In$_{0.81}$Cl$_3$ is isostructural with tetragonal CsTlCl$_3$, which was predicted to be a superconductor with proper doping. The revealed oxidation state of In is not 2+, as the theory predicted, but with a charge-ordering of In$^+$ and In$^{3+}$ on four crystallographic positions in the structure. This is a rare inorganic compound accommodating both In$^+$ and In$^{3+}$ in a nominally double perovskite-derived...
“Cs,In’In’‘In’Cl” structure. The mixed valence of In’ and In’‘ is confirmed by detailed X-ray absorption spectroscopy. Raman spectroscopy demonstrates the In’–Cs–In’‘ active mode with a phonon frequency of 278 cm⁻¹, similar to that of the Tl’–Cs–Tl’’ stretching modes in CsTlCl₃ and CsTl₂HgCl₄. Theoretical calculations conducted on the In deficient CsInCl₃ indicate that it is an indirect band gap semiconductor with a band gap of ~2.27 eV. The large indirect band gap suggests that this nonstoichiometric perovskite-derived phase is not a good candidate for photovoltaic application. The experimental results, however, provide a basis for future theoretical calculations for designing related indanic halide perovskites for potential photovoltaic functions or superconductivity.

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